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Introducing a large polar tetragonal distortion into Ba-doped BiFeO$_3$ by low temperature fluorination


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Introducing a large polar tetragonal distortion into Ba-doped BiFeO₃ by low temperature fluorination

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Abstract

This article reports on the synthesis, crystallographic and magnetic structure of barium-doped BiFeO$_3$ compounds with approximate composition Bi$_{1-x}$Ba$_x$FeO$_{3-x/2}$, as well as of the fluorinated compounds Bi$_{1-x}$Ba$_x$FeO$_{3-x}F_x$ (both with $x = 0.2$, 0.3), prepared by low-temperature fluorination of the oxide precursors using polyvinylidenedifluoride, PVDF. Whereas the oxide compounds were obtained as cubic ($x = 0.2$) resp. slightly tetragonal ($x = 0.3$, $c/a \sim 1.003$) distorted perovskite compounds, a large tetragonal polar distortion was observed for the oxyfluoride compounds ($c/a \sim 1.08$ for $x = 0.2$ and $\sim 1.05$ for $x = 0.3$), being isostructural to tetragonal PbTiO$_3$. Although described differently in previous reports on Ba doped BiFeO$_3$, the observed remanent magnetisation is found to agree well with the amount of BaFe$_{12}$O$_{19}$ only detectable by neutron diffraction and the well-known magnetic properties of BaFe$_{12}$O$_{19}$. The oxyfluoride compounds show G-type antiferromagnetic ordering with magnetic moments lying in the $a/b$-plane.

Keywords

BiFeO$_3$; fluorination; tetragonal distortion; magnetic properties; magnetic structure
1 Introduction

Piezoelectric, particular ferroelectric materials are important for the use in many devices \(^1\), with Pb(Ti,Zr)O\(_3\) being the dominant choice in industrial applications such as ink jet printers, gas igniters, micro-positioning systems, sonar, medical ultrasonic transducers or actuators for fuel injection \(^2,^3\). However, for toxicological reasons there is high interest in making lead-free perovskite based ferroelectric piezoceramics \(^4,^5\) using highly abundant, environmentally friendly, low-cost elements. In this respect various solid solutions of bismuth-based compounds, such as Bi\(_{0.5}\)Na\(_{0.5}\)TiO\(_3\), Bi\(_{0.5}\)K\(_{0.5}\)TiO\(_3\), or BiFeO\(_3\) have been widely studied \(^6-^{10}\). In the case of BiFeO\(_3\), it is of significant interest but generally has an intrinsic conductivity that is too high to be useful \(^11\). In the last decade, this material gained broader interest for its multiferroic properties, which potentially make it attractive in the field of data storage \(^12-^{18}\). Both ferroelectric and antiferromagnetic ordering of BiFeO\(_3\) are stable with Curie and Néel temperatures of 1098 K and 643 K \(^11\).

The structure of BiFeO\(_3\) can be derived from the cubic perovskite structure. Under ambient conditions, the compound crystallizes in the trigonal polar space group \(R3c\) and shows spiral antiferromagnetic ordering \(^19\). It has been reported that the crystal structure can be affected by chemical doping as well as by introducing strain when preparing the material in thin film form. Usually, attempts to replace Bi\(^{3+}\) by a lower valent alkaline earth ion AE\(^{2+}\) (resulting in the formation of vacancies at the same time following a composition of Bi\(_{1-x}\)AE\(_x\)FeO\(_{3-x/2}\) \(^{20-25}\) or by isovalent La\(^{3+}\) (to form Bi\(_{1-x}\)La\(_x\)FeO\(_3\) \(^{26,^27}\), leads to transformation to a (pseudo)cubic perovskite compound with increasing amount of the dopant. It should be noted that although the cubic space group is centrosymmetric, and thus theoretically prohibitive for the adaptation of a ferroelectric polarization, the local symmetry has been shown to be lower for a
variety of compounds \(^{28-32}\). A change from trigonal to tetragonal was reported for the preparation of compounds of the system BiFeO\(_3\) – PbTiO\(_3\), i.e. by co-doping BiFeO\(_3\) on the A and B site with Pb\(^{2+}\) and Ti\(^{4+}\) ions respectively \(^{33}\), a single phase material with a large c/a-ratio of \(~ 1.19\) was observed.

For the preparation of thin films, the substrate and the deposition condition has been shown to be highly influential in determining the kind of distortion which is induced by the perovskite compound. Theoretical calculations predict the formation of a tetragonal \(P4mm\) type distortion only for conditions which induce a high degree of strain for pure BiFeO\(_3\) \(^{14}\). By this approach, a large tetragonal splitting with a c/a-ratio of \(~ 1.25\) can be induced for thin films deposited on SrTiO\(_3\), which is often described as supertetragonal. In addition, it was shown that chemical substitution of Bi\(^{3+}\) for Ba\(^{2+}\) can also introduce a tetragonal distortion in thin films at far lower strains \(^{34}\), and also A-site doping using neodymium has been shown to be successful in this respect \(^{35}\). Furthermore, Iliev et al. reported that supertetragonal BiFeO\(_3\) should be favorable in terms of higher ferroelectric polarization values and simpler switching properties \(^{36}\).

So far, no tetragonal \(P4mm\) type AFeX\(_3\) type phase (A = mainly Bi\(^{3+}\), X = mainly oxygen) has been reported apart from thin film preparations, and computational results indicate that such a phase should be energetically less stable \(^{14}\).

Topochemical manipulations are important reaction types especially in the field of battery type materials \(^{37-39}\) and for switchable material properties \(^{40}\). They are widely studied reaction types for directing structure properties of perovskite type materials \(^{41}\). Among them, low temperature fluorination reactions \(^{42-44}\) are viable tools to influence the composition of the anion sublattice, to alter material properties (such as the introduction of superconductivity \(^{45-47}\)) as well as to study the effect on magnetic structures and behaviour \(^{48, 49}\), but are also interesting from a purely structural point
of view in terms of anion ordering, whose understanding “would be critical if we are to utilize anion substitutions effectively to tailor the properties of materials” 50. Such reactions are usually necessary (instead of simple high temperature reactions, which only work for few compounds for low fluoride contents51-55) due to the high stability of the alkaline earth fluorides AEF₂ and lanthanide oxyfluorides LnOF. A variety of fluorination agents have been used for such topochemical reactions, among them F₂, NH₄F, MF₂ (M = Cu, Zn), XeF₂ (see Ref. 56, 44 for an overview), and it was recently shown that topochemical fluorination reactions can even be performed using electrochemical methods 57. However, for the preparation of iron containing oxyfluoride perovskites polyvinlylidenedifluoride, PVDF 56, has proved to be a viable fluorination agent, and the compounds SrFeO₂F 58, 59, BaFeO₂F (so far known in three different modifications) 60-63, Sr₁ₓBaₓFeO₂F 64 and La₁ₓSrₓFeO₃₋ₓFₓ 65, 66 have all been synthesized using this fluorine containing precursor. For all these systems (and also Fe-containing Ruddlesden-Popper type phases prepared using PVDF 67), the minimum Fe oxidation state was found to be Fe³⁺ and no compounds with higher fluorine contents than the ones needed to obtain single valent Fe³⁺ only have been reported so far (i. e. subsequent further reductive substitution of 1 O²⁻ by 1 F⁻ 44). Therefore, if one wants to modify BiFeO₃ using PVDF, substitution of the Bi³⁺ ion by lower valent alkaline earth ions will be required to make this reaction type applicable.

In this article we report on the first synthesis of tetragonal (P4mm type) Bi₁₋ₓBaₓFeO₃₋ₓFₓ (x = 0.2, 0.3) with a structure similar to PbTiO₃ as a bulk powder, showing a large c/a-ratio of ~1.08 - 1.05. The compounds were synthesized by solid state reaction to prepare the precursor oxides and subsequent topochemical fluorination of the oxides using PVDF at reduced temperatures. So far, these are the first perovskite type ferrite compounds containing only Fe³⁺ on the B-site, and
showing this structural arrangement. In addition, we report on the magnetic structure of the as prepared oxide and oxyfluoride compounds, which show G-type order of the magnetic moments. Furthermore, the identification of the ferrimagnetic impurity phase BaFe$_{12}$O$_{19}$ by means of neutron powder diffraction will likely provide an important new insight into the remanent magnetisation found for Ba-doped BiFeO$_3$, which has been previously described as an intrinsic property of this phase.$^{68,21}$

2 Experimental

2.1 Synthesis of compounds

Approximately 4 g of samples of composition Bi$_{1-x}$Ba$_x$FeO$_{3-x/2}$ (x = 0.2, 0.3) were synthesized by conventional solid state reactions. Stoichiometric amounts of starting powders of BaCO$_3$ (Sigma Aldrich, ≥99%), Bi$_2$O$_3$ (Fluka, > 98 %) and Fe$_2$O$_3$ (Fluka, ≥ 99 %) were ground in a planetary ball mill (Retsch PM 100 CM, 300 rpm, 1 h) using iso-propanol as dispergent. The samples were then heated at temperatures of 850 °C and 880 °C for 15 h, with a regrinding before each further heat treatment. For x = 0.3, the sample was heated a third time at 920 °C for another 15 h, since Ba-richer compounds are known to require higher reaction temperatures.$^{68,69,21}$ All heat treatments were performed in covered alumina crucibles without using an excess of Bi$_2$O$_3$.

Fluorination was achieved by grinding the as-synthesized powders with stoichiometric amounts of PVDF (Aldrich) in a 20 % excess (i.e. 1 : 0.6 molar ratio of x in Bi$_{1-x}$Ba$_x$FeO$_{3-x/2}$F$_x$ : CH$_2$CF$_2$ monomer unit), followed by slowly heating the mixture to a temperature of 370 °C for 15 h again using covered alumina crucibles.
2.2 Diffraction experiments

X-ray powder diffraction (XRD) patterns of the oxide and oxyfluoride compounds were recorded on a Bruker D8 diffractometer with Bragg-Brentano geometry and a fine focus X-ray tube with Cu anode. No primary beam monochromator was attached. A VANTEC detector and a fixed divergence slit (0.1 °) were used. The total scan time was 10 hours for the angular range between 5 and 130° 2θ.

High-temperature X-ray powder diffraction (XRD) patterns of the oxyfluoride compounds were recorded on the same setup (except for using a 0.3 ° fixed divergence slit) using an Anton Paar HTK 1200N High-Temperature Oven-Chamber in a temperature range between 50 and 550 °C with a scan time of 1.5 hours per scan.

Time of flight powder neutron diffraction (NPD) data were recorded on the HRPD high resolution diffractometer at the ISIS pulsed spallation source (Rutherford Appleton Laboratory, UK). 4 g of powdered samples were loaded into 8 mm diameter thin-walled, cylindrical vanadium sample cans and data collected at ambient temperature for 40 µAh proton beam current to the ISIS target (corresponding to ~1 hour beamtime) for the oxide compounds and for 80 µAh proton beam current to the ISIS target (corresponding to ~2 hours beamtime) for the oxyfluoride compounds.

Refinement of the magnetic and nuclear structure of Bi1-xBaxFeO3-xFx and Bi1-xBaxFeO3-x/2 (x = 0.2) was performed with the program TOPAS Academic 5 70, 71 using the NPD data collected in all of the HRPD detector banks 1-3 at room temperature as well as the XRD data.
2.3 Magnetometric measurements
Field-dependent DC susceptibility measurements were performed using a Quantum Design MPMS-XL 5 SQUID magnetometer at 300 K between 0 and 4.8 T.

2.4 Mössbauer measurements
The $^{57}\text{Fe}$ Mössbauer spectrum for Bi$_{0.7}$Ba$_{0.3}$FeO$_{2.7}$F$_{0.3}$ was recorded in standard transmission geometry in constant acceleration mode using a ca. 15 mCi $^{57}\text{Co}/\text{Rh}$ source at room temperature. The data are computed using the WinNormos software by R. A. Brand (WISSEL company). The isomer shift is quoted relative to metallic iron at room temperature.

2.5 Scanning Electron Microscopy (SEM) and Energy Dispersive X-ray Spectroscopy (EDX)
EDX spectra have been recorded for Bi$_{1-x}$Ba$_x$FeO$_{3-x}$F$_x$ ($x = 0.2, 0.3$) using a Philips XL30 FEG scanning electron microscope operating at 30 keV. For EDX analysis the EDAX Genesis system was used and an energy resolution of about 140 eV was applied. The mapped area was of the order of 100 $\mu$m$^2$ and the atomic ratios were determined from the average of the Bi-M, Ba-L, and Fe-K lines, which were averaged over two spots of the sample. The samples were sputtered with approximately 10 nm of Au prior to the measurements.

2.6 Electrical Measurements
The sample powders were cold-isostatically pressed at 300 MPa (KIP 100 E, Paul-Otto Weber GmbH, Remshalden, Germany). Electrodes were applied using either Ag paint (G3692 Acheson Silver Dag 1415, Plano GmbH, Wetzlar, Germany) or Pt paint (Gwent Group Ltd., Pontypool, U.K.) and dried at 100°C for 1 hour. Temperature and frequency-dependent permittivity and impedance measurements were performed using an impedance analyzer (HP 4192A, Hewlett-Packard Co., Palo Alto, USA). For
polarization measurements, the samples were first vacuum infiltrated with a silicone oil (AK 35, Wacker Chemie GmbH, München, Germany), which allowed to apply electric fields up to 6 kV/mm. For these polarization hysteresis measurements, a commercially available setup (aixPES system, aixACCT Systems GmbH, Aachen, Germany) was used. In this commercial system, polarization is measured according to the virtual ground method.

The reliable electrical characterisation proved not to be possible due to the inability to sinter the samples without decomposition: the results from the attempted measurements are detailed in the Supplementary Material.

3 Results and Discussion

3.1 Analysis of the nuclear and magnetic structure of Bi$_{1-x}$Ba$_x$FeO$_{3-x/2}$ ($x = 0.2$, 0.3) at ambient temperature

When synthesizing BiFeO$_3$ type compounds, it is known that the volatility of unreacted Bi$_2$O$_3$ at high temperature can cause the loss of large amounts of this component. Therefore we attempted solid state reaction after using high energy ball milling to produce a reactive mixing of stoichiometric amounts of the oxide / carbonate precursors for the reaction. The ratios of the elements were confirmed to be correct within errors by an EDX analysis (see Table 1) of the fluorinated compounds Bi$_{1-x}$Ba$_x$FeO$_{3-x}$Fx ($x = 0.2$, 0.3) (see section 3.2.1). The cation ratio is unaffected by the fluorination due to the very low reaction temperatures (below 400 °C) and the fact that the fluorination reaction is a topochemical reaction. Although we cannot rule out small deviations from the ideal stoichiometry (e. g. following an equation Bi$_{1-x-d}$Ba$_{x+d}$FeO$_{3-(x+d)/2}$ / Bi$_{1-x-d}$Ba$_{x+d}$FeO$_{3-x-d}$Fx$_{x+d}$), the general conclusions drawn in this article are independent of such small compositional
deviations, and the excess of fluorination agent used is enough to compensate such compositional variations.

Table 1. Results of an EDX analysis of Bi$_{1-x}$Ba$_x$FeO$_{3-x}$Fx ($x = 0.2$, 0.3). *:= $\sigma$ was calculated from the intensity errors. The real $\sigma$ is higher, since other sources of error (e.g. fit of background) would have to be taken into account.

<table>
<thead>
<tr>
<th>Bi$_{1-x}$Ba$<em>x$FeO$</em>{3-x}$Fx</th>
<th>$x = 0.2$</th>
<th>$x = 0.3$</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td>theoretic al</td>
<td>measured ± $\sigma^*$</td>
</tr>
<tr>
<td>Fe-K</td>
<td>1</td>
<td>1 ± 0.02</td>
</tr>
<tr>
<td>Ba-L</td>
<td>0.2</td>
<td>0.22 ± 0.05</td>
</tr>
<tr>
<td>Bi-M</td>
<td>0.8</td>
<td>0.79 ± 0.01</td>
</tr>
<tr>
<td>ratio Bi/Ba</td>
<td>4.00</td>
<td>3.52</td>
</tr>
<tr>
<td>ratio Fe/Bi</td>
<td>1.25</td>
<td>1.27</td>
</tr>
<tr>
<td>ratio Fe/Ba</td>
<td>5.00</td>
<td>4.47</td>
</tr>
</tbody>
</table>

The as synthesized oxide products were obtained as a (pseudo)cubic perovskite compound for $x = 0.2$, resp. a slightly tetragonal distorted perovskite compound for $x = 0.3$ ($c/a \approx 1.003$, only detectable in the high resolution NPD bank 1 data; also see comments on the tetragonal distortion found for Bi$_{0.5}$Ba$_{0.5}$FeO$_{2.75}$ later in this section). In the XRD patterns, a small amount of impurity of either tetragonal β-Bi$_2$O$_3$ ($x = 0.2$, ~ 2 wt-%), which is known to be stabilized by small degrees of doping, and trigonal Bi$_{0.84}$$\delta$Ba$_{0.156}$O$_{1.422}$ ($x = 0.3$, ~4 wt-%) were found, with no indication of further impurities being present. However, although undetectable in the XRD patterns, the presence of the ferrimagnetic compound BaFe$_{12}$O$_{19}$ at between 2 (x = 0.2) to 4 (x = 0.3) wt-% was clearly indicated in the neutron diffraction patterns. Since this phase shows magnetic ordering up to high temperatures ($T_C = 450$ °C), its magnetic reflections were also present in the pattern and could be described using the magnetic structure reported previously, with a magnetic moment of ~ 4 $\mu_B$ per Fe$^{3+}$ ion. The presence of BaFe$_{12}$O$_{19}$ is understandably far more difficult to be
detected in the XRD data, which is reasonably explained by the fact that it contains mainly weaker scatterers compared to the main perovskite phase and also that its main reflection overlaps with the tails of the most intense (1 1 0) perovskite reflection. In contrast, the weaker neutron scatterers Bi$_2$O$_3$ and Bi$_{0.844}$Ba$_{0.156}$O$_{1.422}$ are very hard to detect in the neutron diffraction pattern (compared to phases containing the stronger scattering Fe atom) and their strongest neutron reflections additionally overlap with the perovskite reflections.

For the perovskite compound, magnetic reflections were found in the pattern which can be indexed using a k-vector of $[\frac{1}{2} \ \frac{1}{2} \ \frac{1}{2}]$. Refinement of the magnetic structure of both compounds ($x = 0.2$ and 0.3) indicates G-type antiferromagnetic order with magnetic moments close to $3.74(2) \ \mu_B$ ($x = 0.2$) and $3.70(2) \ \mu_B$ ($x = 0.3$) per Fe atom. This value is similar in magnitude to those reported for other Fe$^{3+}$ containing perovskite compounds at room temperature $^{55, 62, 66, 63, 79}$. Since a hypothetical canting moment would be very small ($< 0.1 \ \mu_B$ per Fe atom) and due to the limited number of magnetic reflections present, such canting of the magnetic moment would be too small to be refined reliably from the NPD data. However, from the absence of further magnetic (superstructure) reflections or satellites, we conclude that the spiral magnetic order of BiFeO$_3$ is indeed destroyed by A-site doping with Ba$^{2+}$ $^{19}$.

No further reflections could be detected for $x = 0.2$ (neither in the XRD nor in the NPD data) apart from the ones belonging to the nuclear and magnetic phases reported above. For $x = 0.3$, one additional reflection at $d \approx 3.199$ Å was detected but could not be assigned to one of the mentioned phases.

From XRD data alone, Khomchenko et al. $^{68, 69, 21, 23}$ and Wang et al. $^{80}$ have reported that Bi$_{1-x}$Ba$_x$FeO$_{3-x/2}$ crystallizes in the trigonal space group $R3c$, being isotypic to the aristotype BiFeO$_3$. With regards to their diffraction patterns and/or reported trigonal
lattice parameters, no superstructure reflections due to the lowering of symmetry from \(Pm-3m\) to \(R3c\) are visible in the patterns shown and also the trigonal lattice parameters appear to be close to pseudocubic, approximately following the simple relation \(c_{R3c} \sim a_{R3c}^*6^{0.5}\). The Bi ion also does not shift far from its position for a pseudocubic arrangement of \((0, 0, \frac{1}{4})\) with \(z \sim 0.254\) (compared to \(z \sim 0.22\) reported in \(^81\)). In contrast, Dachraoui et al. \(^{29}\) report on the formation of a cubic structure for \(\text{Bi}_{0.81}\text{Pb}_{0.19}\text{FeO}_{2.905}\). However, by the use of TEM and from their structural analysis of synchrotron XRD data, it was found that the local symmetry is lower, with shifts of Bi ions along \([1 1 0]\) or \([1 1 1]\), and that the vacancy order cannot be considered to be random. In addition, the cubic space group was also reported for Ca-doped BiFeO\(_3\) by Chen et al. \(^{22}\).

A non-polar tetragonal superstructure was reported for compositions close to \(\text{Bi}_{0.6}\text{Ba}_{0.5}\text{FeO}_{2.75}\) by Boullay et al. \(^{73}\) \(^{74}\) (space group \(P4/mmm\), \(c = 4c_{\text{primitive}}\), \((c/a)_{\text{primitive}} \sim 1.02\)), with superstructure reflections being clearly visible both in the XRD and NPD and showing additional incommensurate order. When preparing the oxides, we often found that a mixture of a tetragonal and a cubic phase was formed, for the first heating instances, with the amount of the cubic phase increasing when heating to higher temperatures. We assume that this indicates a non-random distribution of the A site cations at lower temperatures, with Ba richer regions belonging to the tetragonal phase observed by Boullay et al. and Bi richer regions belonging to the cubic phase. Furthermore, such tetragonal phases are more likely to be found for compounds with increasing Ba content.

Plots of the Rietveld analyses are shown for \(\text{Bi}_{0.8}\text{Ba}_{0.2}\text{FeO}_{2.9}\) and for \(\text{Bi}_{0.8}\text{Ba}_{0.2}\text{FeO}_{2.9}\) in Figure 1 and Figure 2; the structural data are listed for \(\text{Bi}_{0.8}\text{Ba}_{0.2}\text{FeO}_{2.9}\) and for \(\text{Bi}_{0.7}\text{Ba}_{0.3}\text{FeO}_{2.85}\) in Table 2 and Table 3.
For both compounds, we tried to allow for displacement of the Bi and O ion along the [0 0 1], [0 1 1], and [1 1 1] directions. For all the directions, the shift away from the ideal cubic site was relatively small (far smaller compared to what we found for the fluorinated compounds described in section 3.2.1, and also smaller compared to the standard deviation of the refined value), with thermal parameters of the A-site cations Bi and Ba being very high. This is in agreement with the general disorder on this site and could additionally indicate that Bi shifts away from its ideal position due to vacancies in its neighborhood (also see discussion of structural changes upon fluorination in section 3.2.1). Since those vacancies are likely to be randomly (or nearly randomly) distributed (see Mössbauer studies reported in section 3.4), this effect can be described by strong thermal motion for the Rietveld refinement. Both compounds were finally refined in the tetragonal space group $P4mm$, although it must be stated that especially for $x = 0.2$ the exact symmetry of this off-centre shift remains questionable (the difference in lattice parameters is below the resolution limit of the diffraction data for this compound). It is worth mentioning that $P4mm$ is a translationengleiche subgroup of $P4/mmm$, and therefore, also a $P4/mmm$ type structural distortion can be refined using this model (with the ions being then located on the ideal positions). Indeed, no indicative shifts away from the ideal positions could be observed for the Bi, Ba, and O ions, indicating that the assumption of an overall polar model of the compound is not correct, although the local symmetry is very likely to be reduced.
Figure 1. Rietveld analysis of the nuclear and magnetic structure of Bi$_{0.8}$Ba$_{0.2}$FeO$_{2.9}$. HRPD bank 1 (a) and XRD (b) data are shown. HRPD bank2 and bank3 data are provided as Figure S1a and Figure S1b in the Supplementary Material.

Figure 2. Rietveld analysis of the nuclear and magnetic structure of Bi$_{0.7}$Ba$_{0.3}$FeO$_{2.85}$ (unknown reflection at d = 3.199 is marked with *). HRPD bank 1 (a) and XRD (b) data are shown. HRPD bank2 and bank3 data are provided as Figure S2a and Figure S2b in the Supplementary Material.
Table 2. Structural parameters of the compound with nominal composition Bi$_{0.8}$Ba$_{0.2}$FeO$_{2.9}$, space group $P4mm$, $Z = 1$. * := the position of the Fe ion was fixed to 0.5.

<table>
<thead>
<tr>
<th>Atom</th>
<th>Atom type</th>
<th>Site</th>
<th>x</th>
<th>y</th>
<th>z</th>
<th>occ.</th>
<th>$U_{1,1}$</th>
<th>$U_{3,3}$</th>
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<tbody>
<tr>
<td>Bi</td>
<td>Bi$^{3+}$</td>
<td>1a</td>
<td>0</td>
<td>0</td>
<td>-0.03(1)</td>
<td>0.8</td>
<td>0.086(4)</td>
<td>0.059(3)</td>
</tr>
<tr>
<td>Ba</td>
<td>Ba$^{2+}$</td>
<td>1a</td>
<td>0</td>
<td>0</td>
<td>0.06(8)</td>
<td>0.2</td>
<td></td>
<td></td>
</tr>
<tr>
<td>Fe</td>
<td>Fe$^{3+}$</td>
<td>1b</td>
<td>½</td>
<td>½</td>
<td>0.5 *</td>
<td>1</td>
<td>0.006(1)</td>
<td>0.031(3)</td>
</tr>
<tr>
<td>O</td>
<td>O$^{-2}$</td>
<td>1b</td>
<td>½</td>
<td>½</td>
<td>-0.01(1)</td>
<td>0.967</td>
<td>0.119(6)</td>
<td>0.050(6)</td>
</tr>
<tr>
<td>O</td>
<td>O$^{-2}$</td>
<td>2c</td>
<td>0</td>
<td>½</td>
<td>0.52(1)</td>
<td>0.967</td>
<td>0.006(1)</td>
<td>0.149(7)</td>
</tr>
</tbody>
</table>

$a = 3.9853(1)$, $c = 3.9865(1)$

$R_{wp}$ (total) 11.3 $R_{Bragg}$ [%] (XRD) 1.7

GOF (total) 7.2 $R_{exp}$ [%] (bank1) 3.2

Table 3. Structural parameters of the compound with nominal composition Bi$_{0.7}$Ba$_{0.3}$FeO$_{2.85}$, space group $P4mm$, $Z = 1$. * := the position of the Fe ion was fixed to 0.5.

<table>
<thead>
<tr>
<th>Atom</th>
<th>Atom type</th>
<th>Site</th>
<th>x</th>
<th>y</th>
<th>z</th>
<th>occ.</th>
<th>$U_{1,1}$</th>
<th>$U_{3,3}$</th>
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<td>0</td>
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<tr>
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<td>0</td>
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<td>0.3</td>
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<td></td>
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<tr>
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<td>½</td>
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<td>0.028(3)</td>
</tr>
<tr>
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<td>½</td>
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<tr>
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<td>0.95</td>
<td>0.007(1)</td>
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$a = 3.9963(1)$, $c = 4.0032(1)$

$R_{wp}$ (total) 10.7 $R_{Bragg}$ [%] (XRD) 2.7

GOF (total) 7.0 $R_{exp}$ [%] (bank1) 2.9

In addition, we found that the reflections of the perovskite phase showed anisotropic broadening (e. g. the (1 1 1) reflection and higher orders thereof suffer from the lowest broadening, similar to that reported by Dachraoui et al. for Bi$_{0.81}$Pb$_{0.19}$FeO$_{2.905}$ $^{29}$), and this was taken into account by using a model for anisotropic strain broadening utilizing spherical harmonics.

The presence of the ferrimagnetic compound BaFe$_{12}$O$_{19}$ makes a detailed discussion of any findings / changes of magnetic properties very difficult. We have shown here that small amounts of BaFe$_{12}$O$_{19}$ would be undetectable (or at least extremely difficult to detect) using laboratory quality XRD data alone, even for very long measurement times, and that NPD can be used to show the presence of this phase even at these very small levels. However, since this compound can appear during synthesis, we
believe that our findings offer a potential alternative explanation to the canting of spins reported by Khomchenko et al. \cite{68, 69, 21, 23} and Wang et al. \cite{80}, who both used similar synthetic procedures compared to those we report in this article. In the case of the Bi$_{1-x}$Ba$_x$FeO$_{3-x/2}$ compounds reported here (and also of the Bi$_{1-x}$Ba$_x$FeO$_{3-x}$F$_x$ compounds reported in section 3.2), the observed remanent magnetic moments fit very well to the amount of BaFe$_{12}$O$_{19}$ found on evaluation of the NPD data (and also to the magnitude of the magnetic moment reported by Khomchenko et al. \cite{68, 69, 21, 23}, the magnetisation reported by Wang et al. \cite{80} is higher by a factor of \sim4). This is shown by a detailed magnetic characterisation and correlation with the diffraction experiments reported in section 3.5. Keeping further in mind that the largest remanent magnetic moments were reported by Khomchenko et al. for Ba- and Pb-doping (compared to Ca- or Sr-doping) and increasing Ba/Pb content, and that for both compounds the ferrimagnetic phases AFe$_{12}$O$_{19}$ (A = Ba, Pb) are well known, an extrinsic nature of the remanent magnetic moment might also plausibly explain the observation of remanent magnetic moments for A-site doped BiFeO$_3$. Such a presence of small amounts of PbFe$_{12}$O$_{19}$ was already assumed (although this compound was not detected by diffraction experiments) by Dachraoui et al. for Bi$_{0.81}$Pb$_{0.19}$FeO$_{2.905}$ \cite{29}, observing also a small remanent magnetic moment of \sim0.001 $\mu_B$/f.u., similar to what was reported by Khomchenko et al. \cite{68, 69, 21, 23} (\sim0.005 $\mu_B$/f.u.).

3.2 Analysis of the nuclear and magnetic structure of Bi$_{1-x}$Ba$_x$FeO$_{3-x}$F$_x$

(x = 0.2, 0.3) at ambient temperature

3.2.1 Analysis of the nuclear structure of Bi$_{1-x}$Ba$_x$FeO$_{3-x}$F$_x$ (x = 0.2, 0.3)

The X-ray powder diffraction patterns of the oxides in comparison to the oxyfluorides are shown in Figure 3. It was found that a tetragonal distortion with a large c/a-ratio
is introduced into the material upon fluorination. The c/a-ratio of 1.08 for Bi$_{0.8}$Ba$_{0.2}$FeO$_{2.8}$F$_{0.2}$ is also significantly higher than the value of 1.05 found for Bi$_{0.7}$Ba$_{0.3}$FeO$_{2.7}$F$_{0.3}$, agreeing well with the fact that the tetragonal distortion and its magnitude must be related to the ns$^2$ cation Bi$^{3+}$. The fluorination furthermore causes an increase of the cell volume for the oxyfluorides compared to the oxides of 1.4 Å$^3$ (x = 0.2) and 1.1 Å$^3$ (x = 0.3). This change in volume is not affected by change of oxidation state (as often observed for Fe-containing perovskites$^{64, 65}$, but ruled out from the Mössbauer measurements reported in section 3.4 which confirm that the fluorinated compound Bi$_{0.7}$Ba$_{0.7}$FeO$_{2.7}$F$_{0.3}$ contains only Fe$^{3+}$), but is related to an increased anion interaction with the lone-pair (which is additionally accompanied by the tetragonal distortion).

Figure 3. XRD patterns of the oxide and oxyfluoride compounds Bi$_{1-x}$Ba$_x$FeO$_{3-x/2}$ and Bi$_{1-x}$Ba$_x$FeO$_{3-x}$Fx.

Apart from the perovskite compound, a small impurity of BiOF was detected for both compositions, which is likely to be formed from the impurity of tetragonal Bi$_2$O$_3$ (or Bi$_{0.844}$Ba$_{0.156}$O$_{1.422}$) since the respective amounts compare very well. In the neutron diffraction pattern, the small amount of the ferrimagnetic impurity BaFe$_{12}$O$_{19}$ also
present in the precursor oxide was found to be unaffected by the fluorination procedure.

BaFe$_{12}$O$_{19}$ is known for its ferrimagnetic properties. As for the pure oxides, we also found additional reflections which could be attributed to the magnetic scattering from the impurity phase BaFe$_{12}$O$_{19}$ and could be refined using the magnetic structure reported previously.$^7$8

To study the detailed crystallographic (and also magnetic structure, see section 3.2.2), a combined Rietveld analysis of XRD and NPD was performed. No superstructure reflections (apart from the magnetic reflections) or reflections not explained by the BiOF or BaFe$_{12}$O$_{19}$ impurity phases could be observed for $x = 0.2$, indicating that the nuclear structure can be refined on the basis of the smallest primitive perovskite cell with $a_{\text{nuc}}, c_{\text{nuc}} \sim 4 \text{ Å}. For x = 0.3 \text{ (but not for x = 0.2)}$, we found one additional broad and small reflection (Figure 5b) in the bank 2 NPD data at $d \sim 3.415$-3.420 Å (not present in the precursor oxide), which cannot be indexed on the basis of an $(2^{0.5}$ resp. 2) x a, $(2^{0.5}$ resp. 2) x b, (2 resp. 3 resp. 4) x c unit cell, and is also not described by the superstructure reported by Boullay et al. for Bi$_{0.5}$Ba$_{0.5}$FeO$_{2.75}$ (P4/mmm, c = 4*c$_{\text{primitive}}$). However, the fact that no further reflections, e. g. at $d > 4 \text{ Å}$, could be observed, as well as that no further reflections were found in the XRD data, makes an assumption of an increased unit cell implausible and might indicate the presence of a further small amount of a potential decomposition product. For this size of unit cell, two tetragonal translationengleiche subgroups of the cubic space group Pm-3m can be found, namely the centrosymmetric space group P4/mmm and the non-centrosymmetric space group P4mm. Whereas the space group P4/mmm can be found for compounds with cations for which distortion of coordination polyhedron is influenced by the Jahn-Teller effect
(e. g. RbCuF$_3$ \textsuperscript{82}), the polar space group $P4mm$ is well known for perovskite compounds showing ferro- (and therefore also piezo- and pyro-) electric properties (e. g. BaTiO$_3$ \textsuperscript{83}, Pb(Zr,Ti)O$_3$ \textsuperscript{84} and PbTiO$_3$ \textsuperscript{85}).

Comparing the fits to the neutron and X-ray diffraction patterns for both structural models, we found strong indication that the polar space group (compared to a model within the $P4/mmm$ space group allowing for shifts away from the ideal positions) must be considered to be the valid model. This difference of quality of fit for the two different space groups is clearly evident by eye in the XRD pattern and NPD bank 3 pattern (see Supplementary Material, but also the $\Delta R_{wp}(total) = 1\%$ and $\Delta R_{Bragg, individual}(XRD, all NPD banks) \sim 1\%$ are very significant), indicating that the strongest scatterer Bi$^{3+}$ is responsible most for this polar off-center displacement. The compound was therefore refined in the polar space group $P4mm$ and the fit to the diffraction patterns is shown for Bi$_{0.8}$Ba$_{0.2}$FeO$_{2.8}$F$_{0.2}$ in Figure 4 and for Bi$_{0.7}$Ba$_{0.3}$FeO$_{2.7}$F$_{0.3}$ in Figure 5. This is also confirmed applying Hamilton’s R-test \textsuperscript{86} on the $R_{Bragg}$ values for the different models, which shows that the $P4/mmm$ model can be rejected at the 0.005 level (again see Supplementary Material).

![Figure 4. Rietveld analysis of the nuclear and magnetic structure of Bi$_{0.8}$Ba$_{0.2}$FeO$_{2.8}$F$_{0.2}$. HRPD bank 1 (a) and XRD (b). HRPD bank 1 (a) and XRD (b) data are shown. HRPD bank2 and bank3 data are provided as Figure S3a and Figure S3b in the Supplementary Material.](image-url)
Evaluating the widths of the reflections of the X-ray diffraction patterns and their angular dependence, indicate the presence of both microstrain effects and crystallite size effects contribute significantly to reflection broadening. In addition, we also found that the (0 0 1) reflections are significantly broader than the respective (h 0 0) reflections. Therefore, we had to introduce a model which allows for a more flexible refinement of reflection broadening for the two different crystallographic axes (again by the use of spherical harmonics). Furthermore, we estimated a sort of directional microstrain $\varepsilon_{0,a}$ and $\varepsilon_{0,c}$ by using a fit model with multiple perovskite phases with different lattice parameters (which we define as the fraction of the standard deviation of the lattice parameter to the mean lattice parameter) and found that $\varepsilon_{0,c}$ is nearly tripled compared to $\varepsilon_{0,a}$ (see Table 4 and Table 5, absolute coordinates with respect to the centre of mass are given in the Supplementary Material).

It must also be mentioned that we had to include a second fraction of the perovskite compound to refine the diffraction patterns properly. This is likely due to the tetragonal perovskite not only showing broadening due to strain, but that this broadening is also asymmetric, with slightly higher intensities for c/a-ratios closer to 1 compared to c/a-ratios above 1.08 (for $x = 0.2$) and 1.05 (for $x = 0.3$). These fractions are however found to be at a far lower level than the main tetragonal perovskite
phase (i.e. in a ratio of ~1:4 for x = 0.2 and ~1:9 for x = 0.3), and show a significantly smaller c/a ratio of ~ 1.02 for both compositions together with a slightly reduced cell volume by about ~ 1.4 Å³ (x = 0.2) and ~ 0.7 Å³ (x = 0.3) compared to the main fraction. The cell volume of those fractions is more similar to the corresponding oxide phases; however, adding more PVDF and repeating the heating procedure did not result in “elimination” of this second perovskite type phase. Possibly, this asymmetry can be explained by non-ideal statistical distribution of the Ba²⁺ and Bi³⁺ ions and subsequent varying compositions in the grains. For this second phase, the same structural parameters as the main phase were used. Due to the minor influence of this phase and high reflection overlap, it is not possible to reliably refine structural parameters independently. Furthermore, attempts were made to refine the pattern with a R3c symmetry for the minor fraction, however, the obtained fit was significantly worse compared to using two tetragonal fractions (ΔRwp,total ~ 1 %).

The refined structural parameters for Bi₁₋ₓBaₓFeO₃₋ₓFx are listed in Table 4 and Table 5, and a drawing of the crystal structure is exemplarily shown for x = 0.2 in Figure 6 (both compounds are structurally very similar). Bi₁₋ₓBaₓFeO₃₋ₓFx (x = 0.2, 0.3) show a high similarity to the ferroelectric compound PbTiO₃, i.e. the same kind of off-center shift for the ions by a similar magnitude was observed. In PbTiO₃ the lone pair of the ns² cation (Pb²⁺) is known to be directed towards where the Ba ion is located in Bi₁₋ₓBaₓFeO₃₋ₓFx, and so in Bi₁₋ₓBaₓFeO₃₋ₓFx, a similar directional orientation of the lone pair must therefore also be considered to be highly likely (see Figure 6b).
Table 4. Structural parameters of the compound with nominal composition Bi$_{0.8}$Ba$_{0.2}$FeO$_{2.8}$F$_{0.2}$, space group $P4mm$, $Z = 1$. * := the position of the Fe ion was fixed to 0.5.

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Table 5. Structural parameters of the compound with nominal composition Bi$_{0.7}$Ba$_{0.3}$FeO$_{2.7}$F$_{0.3}$, space group $P4mm$, $Z = 1$. * := the position of the Fe ion was fixed to 0.5.

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Figure 6. Refined crystal structure together with thermal ellipsoids for the different ions of Bi$_{0.8}$Ba$_{0.2}$FeO$_{2.8}$F$_{0.2}$ (a). The location of Bi ion (green) in its anion (red) coordination polyhedron together with the mutual position (as found for PbTiO$_3$) of the lone pair indicated by an yellow arrow (b). For a detailed analysis of the lone pair position, DFT based calculations would be required.

The reasons for the introduction of tetragonal distortion on fluorination may relate to the displacive stereochemical activity of the lone pair being somewhat suppressed in
the Ba-doped Bi$_{1-x}$Ba$_x$FeO$_{3-x/2}$ oxide compounds, but reactivated in the fluorinated compounds. It is known in the literature that the lone pair of the Bi$^{3+}$ ion occupies approximately as much space as an oxide or fluoride anion in the crystal. Therefore, on introduction of vacancies in the anion sublattice, the lone pair is likely to point towards such a vacancy for space reasons. This was already well investigated in a DFT study by Walsh et al. on the different modifications of Bi$_2$O$_3$; for the oxygen disordered $\delta$-modification of Bi$_2$O$_3$ it was found that the ns$^2$ cation forms a lone pair, pointing towards the oxygen vacancy i.e. the “region of space” in the crystal lattice. For perovskite compounds which show a random (or nearly random) distribution of the vacancies in the anion sublattice (as found for Bi$_{1-x}$Ba$_x$FeO$_{3-x/2}$), a pseudocubic arrangement will have to form and a uniform displacive ns$^2$ ion induced distortion will be simply suppressed.

On “refilling” of the anion sublattice with fluoride ions by substituting 1 O$^{2-}$ for 2 F$^-$, the unidirectional displacive effect of the Bi$^{3+}$ ion can be recovered over a large area of the crystallite (O$^{2-}$ and F$^-$ only differ by about 0.06 Å for their ionic radii), since the whole anion sublattice is now occupied by anions. This is in agreement with the increase of cell volume due to fluorination, since the additional anions will require the space which has previously been occupied by the lone pair in the oxide compounds. However, such an increase in cell volume does not necessarily accompany fluorination of Fe-containing perovskite compounds (where the average Fe oxidation state is maintained). For example, the vacancy ordered monoclinically distorted perovskite compound BaFeO$_{2.5}$ reported by Clemens et al. shows a decrease in cell volume on fluorination to BaFeO$_2$F. In contrast, the brownmillerite type SrFeO$_{2.5}$ shows an increase in cell volume on fluorination to SrFeO$_2$F. These differences are explained by the differences in structural distortions in BaFeO$_{2.5}$ and SrFeO$_{2.5}$. 

24
Furthermore, if we consider that the volume changes per fluoride ion (-1.02 Å³/F⁻ for BaFeO₂.₅ and 0.65 Å³/F⁻ for SrFeO₂.₅) are significantly smaller than for the fluorination of Bi₁₋ₓBaₓFeO₃₋ₓFx (7 Å³/F⁻ for x = 0.2 and 3.7 Å³/F⁻ for x = 0.3), we can conclude that the space of the vacancy is likely to have been occupied by the lone-pair of Bi³⁺ in the oxide compounds, which is well known to act as a pseudo-anion ⁹¹.

Although the ns⁰ cation Ba²⁺ does not participate in the off-center displacement and shifts towards the “ideal” position 0, 0, 0 in relation to the position of the Fe atom, the overall polar distortion remains stable in the fluorinated compounds. It must be mentioned at this point that a similar quality of fit was observed when fixing the Ba and Bi ion to the same position which is refined to 0, 0, ~ -0.07 then, without significant change of the positions of the other ions. However, since the refinement proved to be stable and converged properly using a split site for this compound (which would chemically make sense), we have given the split parameters in Table 4 and Table 5.

The adaption of the tetragonal distortion is in contrast to what was found for La-doped BiFeO₃ compounds (Bi₁₋ₓLaₓFeO₃; isovalent substitution of Bi³⁺ by La³⁺) for which a transformation towards a pseudocubic structure is obtained for a far lower degree of doping (around x = 0.1 deduced from the diffraction patterns shown in ²⁷). The difference in ionic radii between La³⁺ and Ba²⁺ is large (1.36 vs. 1.61 Å for 12-fold coordination ⁹⁰) and it has already been shown that structural distortions found in the series La₁₋ₓSrₓFeO₃₋ₓFx (0 ≤ x ≤1, also monovalent Fe³⁺ for all compositions) for increasing La content arise from the fact that the La³⁺ ion tries to lower its effective coordination number ⁶⁵, ⁶⁶. In contrast, a high thermal displacement parameter of the Fe atom was observed for BaFeO₂F, indicating that a local polar displacement is likely in this compound, although an overall cubic perovskite
structure was observed. Therefore, the larger Ba$^{2+}$ and its compatibility with local 12-fold coordination might facilitate the formation of the polar tetragonal distortion for the compounds Bi$_{1-x}$Ba$_x$FeO$_{3-x}$F$_x$, whereas local structural distortions for La$^{3+}$ would just be too strong to maintain the stereochemical activity of Bi$^{3+}$ over larger crystallite sizes.

There also appears to be a clear difference for the Bi$_{1-x}$Ba$_x$FeO$_{3-x}$F$_x$ compounds (x = 0.2, 0.3) compared to PbFeO$_2$F, for which the larger F content and associated disorder of the O$^{2-}$ and F$^{-}$ ions might be prohibitive for the adoption of a uniform polar distortion (i.e. the compound is found in the cubic perovskite structure with strong displacement of the A site cations towards the twelve possible [1 1 0] like directions). So far, it is not entirely clear why PbFeO$_2$F does not show stereochemical activity of the ns$^2$ cation Pb$^{2+}$.

From the shape of the thermal displacement ellipsoids shown in Figure 6a, a slight local tilting of the octahedra is also indicated for Bi$_{1-x}$Ba$_{1-x}$FeO$_{3-x}$F$_x$ (x = 0.2, 0.3) by the disc like shape of the thermal ellipsoid of the O/F1 ion, and blocks with different tilting in the a/b-plane might be present. This would also be in good agreement with a tolerance factor of ~ 0.92-0.94 calculated from the average ionic radii reported in (estimating the radius of Bi$^{3+}$ to 1.36 Å for 12 fold coordination by interpolation). However it is worth mentioning that such a disc like thermal ellipsoid could also arise from the disorder on the A site and shifts of the oxygen ion away from the lone pair of Bi$^{3+}$ towards a Ba$^{2+}$ ion in the same plane. The slightly rod like shape of the thermal ellipsoid of the Fe atom can be explained by a partial split position: for octahedra surrounded by a higher number of Ba$^{2+}$ ions, the Fe ion is more likely to shift towards the center of the octahedron, whereas for octahedra surrounded by a higher number
of Bi$^{3+}$ ions, a shift away from this center towards one corner of the octahedron would be plausible.

Additionally, the direction dependent broadening of the reflections is also in good agreement with the assumption of the lone pair pointing along the c-axis. Since the Bi$^{3+}$ and the Ba$^{2+}$ ion strongly differ in the shape of their electron shell (i. e. rod like vs. sphere like), this would cause a stronger fluctuation of the lattice plane distances along the direction in which the lone pair is pointing at, namely the c-axis for Bi$_{1-x}$Ba$_x$FeO$_{3-x}$F$_x$.

It might be worth mentioning that attempts to synthesize oxyfluorides with a lower Ba-content (i. e. lower values of x) did not result in the formation of phase pure compounds although the precursor oxides appeared to be single phase perovskite compounds. Instead, mixtures of two perovskite type phases (trigonal $R3c$ and tetragonal $P4mm$) were observed. Since no single phase products could be obtained, those compositions haven’t been investigated further.

3.2.2 Analysis of the magnetic structure of Bi$_{1-x}$Ba$_x$FeO$_{3-x}$F$_x$ (x = 0.2, 0.3)

Analysis of the magnetic reflections showed that they are compatible with a k-vector of $[\frac{1}{2} \frac{1}{2} \frac{1}{2}]$. Among the maximal magnetic subgroups of $P4mm$ allowing for this k-vector (determined using the program MAXMAGN$^{92}$ on the Bilbao Crystallographic Server$^{93-95}$), only the groups 108.238 and 46.246 are compatible with a magnetic moment on the Fe atoms. The magnetic space group 108.238 is compatible with a G-type alignment of the magnetic moments along the c-axis, whereas 46.246 is compatible with G-type alignment of the magnetic moment along one of the crystallographic axes a or b. However, since the nuclear cell shows tetragonal symmetry only an overall magnetic moment lying in the a/b-plane can be determined
from powder diffraction data\textsuperscript{96}. Both groups were tested and a proper fit of the reflections could only be obtained for the space group \textit{46.246} (\textit{lma2}), with an overall magnetic moment of 3.7(1) $\mu_B$ per Fe atom. This magnetic moment is again in good agreement with what has been observed for other G-type ordered Fe$^{3+}$ containing perovskites at room temperature\textsuperscript{55, 62, 66, 63, 79}. Furthermore, the alignment of the magnetic moment within the a/b-plane was also found by Boullay et al.\textsuperscript{73} for tetragonal Bi$_{0.5}$Ba$_{0.5}$FeO$_{2.75}$ (\textit{P4/mmm}, $c = 4c_{\text{prim.}}$). None of the maximal subgroups of $P4mm$ with $k = [\frac{1}{2} \frac{1}{2} \frac{1}{2}]$ allow for the presence of a permanent magnetic moment due to canting. We believe that the occurrence of the impurity BaFe$_{12}$O$_{19}$ explains the magnetic moment quite well (see section 3.5), but if such a permanent magnetic moment would exist, it could either lie along the c-axis or even within the a/b-plane. However, this is not indicated in fits using the magnetic space group 1.1 and applying proper constraints to the directions of the magnetic moments between different crystallographic sites (the canting moment would just be too small to be refined reliably), although such a very small canting moment cannot be ruled out entirely by the analysis of the NPD data.

### 3.3 High temperature X-ray diffraction experiments

High temperature X-ray diffraction experiments were performed to determine the decomposition and phase transition behavior of the as prepared oxyfluorides. We found that on heating the compounds to a temperature of approximately 400 °C, the tetragonal polar distortion remains stable and the c/a-ratio also stays nearly constant. On cooling down the compounds to room temperature when not heated above 400 °C, we found that the lattice parameters match very well to what was found during the heating procedure (see Figure 7).
Figure 7. Temperature dependence of the tetragonal lattice parameters a and c as well as the cube root of the cell volume $V^{1/3}$ for heating Bi$_{1-x}$Ba$_x$FeO$_{3-x}$F$_x$ to 400 °C.

When heating the compound to temperatures above 400 °C, a change from tetragonal towards cubic was found. This structural change is accompanied by the formation of BaF$_2$, which is a well-known decomposition product for BaFeO$_2$F type compounds $^{64, 62, 63}$ (also see end of this paragraph). On cooling down from temperatures $\geq 450$°C, the tetragonal distortion was not found to be reappearing, highlighting that the use of topochemical reactions is inalienable for the preparation of oxyfluorides for compounds of the system Bi$_{1-x}$Ba$_x$FeO$_{3-x}$F$_x$ (see Supplementary Material for the pattern of Bi$_{0.7}$Ba$_{0.3}$FeO$_{2.7}$F$_{0.3}$ heated at 550 °C in comparison to undecomposed Bi$_{0.7}$Ba$_{0.3}$FeO$_{2.7}$F$_{0.3}$). The as formed cubic perovskite (respectively slightly distorted perovskite) furthermore shows a similar lattice parameter compared to the precursor oxides Bi$_{1-x}$Ba$_x$FeO$_{3-x}/2$.

We conclude that the polar tetragonal distortion is therefore very stable for the oxyfluorides Bi$_{1-x}$Ba$_x$FeO$_{3-x}$F$_x$, with a Curie temperature $> 770$ K. Structurally similar compounds (e. g. PbTiO$_3$, $T_c = 760$ K) show a clear decrease of the $c/a$-ratio when approaching their Curie temperature (see e. g. $^{97}$), and the fact that this is not
observed for Bi$_{1-x}$Ba$_x$FeO$_{3-x}$F$_x$ might indicate that the tetragonal distortion is even further stabilized for those compositions.

The decomposition of the compounds at elevated temperatures and the subsequent analysis of the decomposition products can be used to prove that incorporation of F into the perovskite lattice was indeed successful (as was already shown in previous articles $^{64, 65, 62, 63}$). The amount of BaF$_2$ agrees within errors with the amount of F$^-$ assumed in Bi$_{1-x}$Ba$_x$FeO$_{3-x}$F$_x$ (e.g. ~ 7 wt-% for Bi$_{0.8}$Ba$_{0.2}$FeO$_{2.8}$F$_{0.2}$). This, together with a completely filled anion sublattice indicated from the neutron data and the presence of single valent Fe$^{3+}$ from the Mössbauer spectroscopic characterization (see section 3.4) provides strong evidence for a composition close to Bi$_{1-x}$Ba$_x$FeO$_{3-x}$F$_x$ is correct for the as prepared oxyfluoride compounds. On decomposition at 700 °C for 5 min, we also found the formation of a rhombohedral ($R3c$) together with a cubic ($Pm-3m$) perovskite phase (in the ratio of ~ 5:4). This indicates that the extraction of Ba due to the formation of BaF$_2$ also causes readoption of the parent rhombohedral phase for Bi-richer regions. The ratio of the cations was confirmed by EDX spectroscopy (within the errors of the method) for both compounds Bi$_{1-x}$Ba$_x$FeO$_{3-x}$F$_x$ ($x = 0.2, 0.3$), see Table 1.

### 3.4 Mössbauer spectroscopy experiments of Bi$_{0.7}$Ba$_{0.3}$FeO$_{2.7}$F$_{0.3}$

$^{57}$Fe Mössbauer spectroscopy was measured on the compound with highest fluorine content Bi$_{0.7}$Ba$_{0.3}$FeO$_{2.7}$F$_{0.3}$. A single sextet (see Figure 8) with an isomer shift characteristic for iron in trivalent oxidation state was found (see Table 6), and the finding of a sextet agrees well with the magnetic structure determined via neutron powder diffraction. However, to obtain a proper fit of the spectrum, a Gaussian type distribution of magnetic fields as well as quadrupole interaction parameters had to be assumed. Such an assumption is reasonable, since due to the disorder of ions in the
compound (e. g. Bi/Ba and O/F) local variations in the exact kind of ion environments will have to occur, and the fact that the quadrupole interaction parameter shows a broader distribution (compared to the distribution of the magnetic hyperfine field parameter) agrees well with this observation.

The magnetic hyperfine field parameter of 49.5 T is comparable to what was found for other magnetically ordered iron containing perovskite compounds with Fe$^{3+}$ being in corner sharing octahedral coordination (e. g. SrFeO$_2$F$^{59}$, 3C-BaFeO$_2$F$^{61}$, 6H-BaFeO$_2$F$^{63}$, 15R-BaFeO$_2$F$^{62}$). Furthermore, the structural characterisation reported in the previous sections can also be validated by comparing the spectrum of Bi$_{0.7}$Ba$_{0.3}$FeO$_{2.7}$F$_{0.3}$ to data of similar compounds reported in literature. Batuk et al. $^{31}$
recorded Mössbauer data for Sr-doped PbFeO\(_{2.5}\). Such lead containing perovskite ferrites show crystallographic shearing in supposedly lead richer regions\(^{30-32}\), which results in the formation of edge sharing between neighbouring Fe-polyhedra. This edge sharing is known to lower the magnetic hyperfine field interaction (e.g. to about 32 T\(^{31}\) due to increased covalent interactions between the Fe and the anions, or due to possibly increased covalent Fe-Fe interactions between face sharing octahedra as found in 6H- and 15R-BaFeO\(_2\)F\(^{62, 63}\)). Crystallographic shearing can therefore be practically ruled out for the fluorinated compounds with tetragonal distortion, since no such lowered magnetic hyperfine field parameters have been found here. This holds also for the oxides due to the topochemical nature of the fluorination reaction. In addition, we could not find any doublet contributions in the Mössbauer spectrum, as was found for Bi\(_{0.81}\)Pb\(_{0.19}\)FeO\(_{2.905}\)\(^{29}\), or a second sextet as was found for Bi\(_{0.8}\)Pb\(_{0.2}\)FeO\(_{2.9}\), which both had been assigned to Fe with lower coordination numbers. Again, this all agrees well with a fully occupied anion lattice for the fluorinated compounds.

3.5 Magnetic characterisation of Bi\(_{1-x}\)Ba\(_x\)FeO\(_{3-x/2}\) and Bi\(_{1-x}\)Ba\(_x\)FeO\(_{3-x}\)F\(_x\)

Field dependent measurements of the magnetisation are shown and discussed for Bi\(_{0.8}\)Ba\(_{0.2}\)FeO\(_{2.9}\) and Bi\(_{0.8}\)Ba\(_{0.2}\)FeO\(_{2.8}\)F\(_{0.2}\) (see inlay of Figure 9) and show a small remanent magnetisation. Both compounds were shown by neutron diffraction to contain small amounts of the impurity phase BaFe\(_{12}\)O\(_{19}\) of 2.0 wt-% for the oxide and 1.7-wt-% for the oxyfluoride. BaFe\(_{12}\)O\(_{19}\) is reported to possess a remanent magnetic moment of \(\sim 20 \mu_B / \text{f.u.} \) (at 0 K)\(^{98}\), with a high Curie temperature of 740 K. This magnetic moment could also be confirmed by the magnetic reflections of this phase present in the NPD data. We tried to correlate the observed remanent magnetisation with the amount of BaFe\(_{12}\)O\(_{19}\) and its known magnetic properties (see Figure 9), and
found that they agree very well within errors (which mainly arise from the quantification of this phase). This explains why the curves of the oxide and oxyfluoride are more similar for the inlay (the ideal molar masses of the oxide and oxyfluoride differ to approximately 1 %) compared to the main figure, for which the magnetisation had to be divided by the weight fractions of 2.0 respectively 1.7 wt-% determined for BaFe₁₂O₁₉. No canting of the magnetic moment of the perovskite compound needs to be assumed to explain the observed magnetic moments, and this is also in agreement with the analysis of the magnetic structures of the oxide as well as the oxyfluoride compounds reported in sections 3.1 and 3.2.

![Figure 9. Field dependent measurements of the magnetisation of Bi₀.₈Ba₀.₂FeO₂.₉ and Bi₀.₈Ba₂.₀₂FeO₂.₉₂F₀.₂ (inlay), measured magnetisation correlated with the amount of BaFe₁₂O₁₉ determined by quantitative phase analysis of the NPD data.](image)

We also found an increase for the magnetisation for increasing Ba content. This is in agreement that for such increase of Ba-content, an increase of the amount of BaFe₁₂O₁₉ was found (~ 4 wt-%), and can be explained by the fact that the samples had to be heated to higher temperatures for obtaining a cubic perovskite phase with nearly random distribution of Bi/Ba ions.
4 Summary

In this article we have shown that tetragonal PbTiO$_3$-type BiFeO$_3$ can be stabilized by Ba-doping of the Bi-site and subsequent topochemical fluorination using PVDF. The as prepared oxyfluoride compounds Bi$_{1-x}$Ba$_x$FeO$_{3-x}$F$_x$ contain single valent Fe$^{3+}$ and show a large tetragonal distortion with a $c/a$-ratio of 1.08-1.05 (for $x = 0.2/0.3$) and crystallize in the polar space group $P4mm$. The compounds are isostructural to Pb(Ti,Zr)O$_3$ compounds, with similar magnitudes for the off-centre displacement of the Fe ions in the octahedra as well as of the ns$^2$ cation Bi$^{3+}$. This is the first time that chemical doping was found to induce a polar tetragonal distortion in non-stressed BiFeO$_3$ compounds, which has been assumed to potentially improve switching kinetics of the compound $^{36}$. The fluorinated compounds show G-type alignment of the magnetic moments in the a/b-plane. The remanent magnetisation found is in agreement with the amount of impurity of BaFe$_{12}$O$_{19}$, which could only be detected and quantified using neutron powder diffraction. Therefore, the intrinsic nature of the magnetic moment due to A-site doping by Ba$^{2+}$ assumed in previous publications $^{80, 99, 68, 21, 25}$ might be questionable.

Currently techniques are being investigated to fluorinate thin films of Ba-doped BiFeO$_3$, which might then be more suitable for electrical characterization for these thermodynamically unstable compounds. The fluorination of thin epitaxially grown films of SrFeO$_{3-\delta}$ was recently reported by Moon et al. $^{100}$ and Katayama et al. $^{101}$ by the use of PVDF and we think that this method might be, in principle, applicable to a broad range of A-site doped Bi$_{1-x}$AxFeO$_{3-y}$ compounds. For such dense films, the evaluation of electric properties might then become feasible.
5 Supporting Information

Supporting information is provided for the comparison of fit models using the space groups $P4mm$ and $P4/mmm$, details for Hamilton’s R-test comparing the structural models of $\text{Bi}_{0.7}\text{Ba}_{0.3}\text{FeO}_{2.7}\text{F}_{0.3}$ with space groups $P4mm$ and $P4/mmm$, and the decomposition of $\text{Bi}_{0.7}\text{Ba}_{0.3}\text{FeO}_{2.7}\text{F}_{0.3}$ at 550 °C. Attempts to determine the electric properties of $\text{Bi}_{1-x}\text{Ba}_x\text{FeO}_{3-x/2}$ and $\text{Bi}_{1-x}\text{Ba}_x\text{FeO}_{3-x}\text{F}_x$ are also included there. Structural data for the compounds $\text{Bi}_{1-x}\text{Ba}_x\text{FeO}_{3-x/2}$ and $\text{Bi}_{1-x}\text{Ba}_x\text{FeO}_{3-x}\text{F}_x$ ($x = 0.2$, 0.3) are provided as cif-files. This material is available free of charge via the Internet at http://pubs.acs.org

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7 References

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Figure Captions

Figure 1. Rietveld analysis of the nuclear and magnetic structure of Bi$_{0.8}$Ba$_{0.2}$FeO$_{2.9}$. HRPD bank 1 (a) and XRD (b) data are shown. HRPD bank2 and bank3 data are provided as Figure S1a and Figure S1b in the Supplementary Material.

Figure 2. Rietveld analysis of the nuclear and magnetic structure of Bi$_{0.7}$Ba$_{0.3}$FeO$_{2.85}$ (unknown reflection at d = 3.199 is marked with *). HRPD bank 1 (a) and XRD (b) data are shown. HRPD bank2 and bank3 data are provided as Figure S2a and Figure S2b in the Supplementary Material.

Figure 3. XRD patterns of the oxide and oxyfluoride compounds Bi$_{1-x}$Ba$_x$FeO$_{3-x/2}$ and Bi$_{1-x}$Ba$_x$FeO$_{3-x}$Fx.

Figure 4. Rietveld analysis of the nuclear and magnetic structure of Bi$_{0.8}$Ba$_{0.2}$FeO$_{2.8}$F$_{0.2}$. HRPD bank 1 (a) and XRD (b).

Figure 5. Rietveld analysis of the nuclear and magnetic structure of Bi$_{0.7}$Ba$_{0.3}$FeO$_{2.7}$F$_{0.3}$ (unknown reflection at d = 3.415-3.420 is marked with *). HRPD bank 1 (a) and XRD (b) data are shown. HRPD bank2 and bank3 data are provided as Figure S4a and Figure S4b in the Supplementary Material.

Figure 6. Refined crystal structure together with thermal ellipsoids for the different ions of Bi$_{0.8}$Ba$_{0.2}$FeO$_{2.8}$F$_{0.2}$ (a). The location of Bi ion (green) in its anion (red) coordination polyhedron together with the mutual position (as found for PbTiO$_3$) of the lone pair indicated by an yellow arrow (b).

Figure 7. Temperature dependence of the tetragonal lattice parameters a and c as well as the cube root of the cell volume $V^{1/3}$ for heating Bi$_{1-x}$Ba$_x$FeO$_{3-x}$Fx to 400 °C.

Figure 8. $^{57}$Fe Mössbauer spectrum recorded from Bi$_{0.7}$Ba$_{0.3}$FeO$_{2.7}$F$_{0.3}$ at room temperature.

Figure 9. Field dependent measurements of the magnetisation of Bi$_{0.8}$Ba$_{0.2}$FeO$_{2.9}$ and Bi$_{0.8}$Ba$_{0.2}$FeO$_{2.8}$F$_{0.2}$ (inlay), measured magnetisation correlated with the amount of BaFe$_{12}$O$_{19}$ determined by quantitative phase analysis of the NPD data.
Ba-doping of BiFeO$_3$ followed by low-temperature fluorination to form compounds of composition Bi$_{1-x}$Ba$_x$FeO$_{3-x}$F$_x$ was found to introduce a large tetragonal polar distortion with PbTiO$_3$-type structure.