UNIVERSITY OF BIRMINGHAM University of Birmingham Research at Birmingham

Hybrid polymer networks of carbene and thiol ene

Djordjevic, Ivan; Wicaksono, Gautama; Singh, Juhi; Singh, Manisha; Ellis, Elizabeth G.; Alraddadi, Maher; Dove, Andrew; Steele, Terry W. J.

DOI 10.1016/j.eurpolymj.2022.111502

License: Creative Commons: Attribution-NonCommercial-NoDerivs (CC BY-NC-ND)

Document Version Peer reviewed version

Citation for published version (Harvard): Djordjevic, I, Wicaksono, G, Singh, J, Singh, M, Ellis, EG, Alraddadi, M, Dove, A & Steele, TWJ 2022, 'Hybrid polymer networks of carbene and thiol ene', *European Polymer Journal*, vol. 178, 111502. https://doi.org/10.1016/j.eurpolymj.2022.111502

Link to publication on Research at Birmingham portal

General rights

Unless a licence is specified above, all rights (including copyright and moral rights) in this document are retained by the authors and/or the copyright holders. The express permission of the copyright holder must be obtained for any use of this material other than for purposes permitted by law.

•Users may freely distribute the URL that is used to identify this publication.

•Users may download and/or print one copy of the publication from the University of Birmingham research portal for the purpose of private study or non-commercial research. •User may use extracts from the document in line with the concept of 'fair dealing' under the Copyright, Designs and Patents Act 1988 (?)

•Users may not further distribute the material nor use it for the purposes of commercial gain.

Where a licence is displayed above, please note the terms and conditions of the licence govern your use of this document.

When citing, please reference the published version.

Take down policy

While the University of Birmingham exercises care and attention in making items available there are rare occasions when an item has been uploaded in error or has been deemed to be commercially or otherwise sensitive.

If you believe that this is the case for this document, please contact UBIRA@lists.bham.ac.uk providing details and we will remove access to the work immediately and investigate.

1	Hybrid polymer networks of carbene and thiol ene
2	Ivan Djordjevic ¹ , Gautama Wicaksono ¹ , Juhi Singh ^{2,3} , Manisha Singh ¹ , Elizabeth G. Ellis ¹ ,
3	Maher A. Alraddadi ⁴ , Andrew P. Dove ⁴ and Terry W.J. Steele ¹
4	
5	¹ School of Materials Science and Engineering (MSE), Nanyang Technological University,
6	Singapore 639798.
7	² School of Chemical and Biomedical Engineering, Nanyang Technological University,
8	Singapore 637457.
9	³ NTU Institute for Health Technologies, Interdisciplinary Graduate Program, Nanyang
10	Technological University (NTU), Singapore 637335.
11	⁴ School of Chemistry, University of Birmingham, Edgbaston, Birmingham B15 2TT, United
12	Kingdom
13	*Corresponding author: Terry W. J. Steele (e-mail: <u>wjsteele@ntu.edu.sg</u>)
14	
15	
16	
17	
18	
19	
20	
21	
22	
23	
24	

ABSTRACT

26	Thiol/ene-based resorbable elastomers display tough elongation but lack adhesion to soft
27	tissues. Carbene-based bioadhesives (e.g. CaproGlu) allow soft tissue adhesion, but the
28	covalent crosslinks limit extensibility after photoactivation. Herein thiol/ene resorbable
29	elastomers are combined with a carbene bioadhesive into a 3-component hybrid network by
30	exploiting tunable photoactivation of each macromolecule independently or simultaneously.
31	Dual crosslinking was monitored by photorheometry, where 405 nm initiates formation of a
32	thiol/ene elastomeric network, followed by 365 nm activation of diazirine-grafted
33	polycaprolactone tetrol (CaproGlu). Dynamic shear moduli, gelation point, elongation at
34	break, and lap shear stress of the hybrid polymer network are evaluated with respect to
35	absorbed light energy dose. Surface-exposed unreacted CaproGlu enables adhesion of the
36	hybrid network to various substrates, as well as intermolecular crosslinking within the
37	transparent matrix. The network morphology and functional group conversion is evaluated
38	through scanning electron microscopy and infrared spectroscopy, respectively. For the first
39	time, we demonstrate hybrid thiol/ene/diazirine double sided bioadhesives with tunable
40	dynamic moduli in the range of 10-800 kPa and 160 kPa lap-shear adhesion strength.
41	
42	Keywords: Hybrid polymer network, diazirine-grafted polycaprolactone; light curing;
43	crosslinked elastomer; double sided adhesive.
44	
45	
46	
47	
48	
49	
50	
51	
52	
53	
54	
55	
55 56	

58 **1. INTRODUCTION**

Hybrid polymer networks consist of two or more entangled polymers which homogeneously 59 build the polymer system by physical and/or covalent crosslinks [1]. The strategy of 60 combining the properties of individual polymers could result in precise tuning of hybrid 61 material for targeted applications [2]. One of the obvious examples of hybrid networks are 62 "interpenetrated polymer network" (IPN) systems that have pushed the limits of what is 63 capable for viscoelastic elastomers, such as elongation that exceeds 2000% and resilience that 64 65 is unmatched with typical rubber networks [3]. Hybrid IPN materials can be produced both in hydrogel [4] or solvent free forms [5]. Careful selection of the type of crosslinking chemistry 66 67 provides the strategy of design and control over unique material performance characteristic to hybrid polymer networks [6]. In applications such as wound management, tissue sealing and 68 69 reconstruction, tissue adhesion plays a significant role. There are a number of hybrid materials reported to adhere to tissues. Recent examples include allyl-functionalized 70 71 branched polymers mixed with tri-thiol crosslinking component [7], *in situ* forming multimonomer acrylate IPN hydrogel tissue patch [8], polyacrylamide/alginate hybrid hydrogels 72 [9] and two-component adhesive, composed by two different *p*-hydroxyphenyl-grafted 73 polymers: chitosan and polyethylene glycol (PEG) activated by hydrogen peroxide and 74 horseradish peroxidase [10]. Most of the bioadhesive systems (single components or hybrid 75 network) rely on interfacial bonding realized by acrylate crosslinking and are either limited to 76 77 topical use (i.e. cyanoacrylate) [11] or result in low adhesion strength (i.e. IPN hydrogels; adhesion strength ~20 kPa) [8]. Another type of tissue adhesion is by physical interaction (i.e. 78 hydrogen or ionic bonds) where adhesion strength could be compromised by hydrolysis or 79 changes in local pH values [12]. Current unmet clinical needs require bioadhesives that 80 deviate from 2-part chemical curing designs. One-pot stimuli-sensitive crosslinking systems 81 82 are sought with specific design parameters, such as: (i) solvent-free liquid resins; (ii) benign light activation energy that would yield rapidly gelling biomaterials; (iii) improved 83 84 crosslinking chemistry to provide model systems for investigations of hybrid network adhesives. On-demand materials are sought for tissue adhesives with sufficient 85 adhesive/cohesive strength to replace mechanical fixation methods based on sutures or 86 87 staples.

88 Using activated esters presents another strategy for interfacial covalent bonding between

89 hybrid adhesive and surface amines on tissue substrates [13-15]. Although some important

90 advances have been made in the field with N-hydroxysuccinimide (NHS)-grafted

biomaterials, it should be noted that adhesion strength is dependent on concentration of 91 surface amines that can vary for different tissues, causing modulation in adhesive 92 performance [16]. Another potential drawback of NHS grafting technology is that requires 93 dehydrated tissue prior to adhesive interaction, which often includes relatively complicated 94 macromolecular designs [12, 13]. Carbene-based bioadhesives are under development to 95 surpass current limitations of commercial tissue adhesives. Diazirine-grafted 96 polycaprolactone polyol (CaproGlu) [17] was designed to be solvent-free liquid precursor for 97 rapid light-induced gelation (both visible and UV light) and to be miscible with organic 98 99 additives [18]. The hydrophobic nature of liquid CaproGlu allows solvent-free dissolution of other hydrophobic liquid polymers to yield hybrid network. Herein, CaproGlu's miscible 100 nature is exploited to dissolve thiol/alkene and thiol/alkyne precursors into hybrid polymer 101 102 network (Figure 1).



103

Figure 1. 3-component double sided adhesive hybrid polymer network design by solvent-free
mixture of (top), 4-functional thiol, PEG-diacrylate and diazirine-grafted polycaprolactone
tetrol (CaproGlu); (bottom) syringable hybrid network undergoes liquid-to-solid transition
upon activation with visible light (405 nm – Step 1) followed by diazirine-to-carbene
activation (UVA light at 365 nm – Step 2) that results in interfacial covalent attachment of
adhesive gel with solid surface.

- 111 Polyethylene glycol-diacrylate (Ene), pentaerythritol tetrakis (3-mercaptopropionate) (SH), 2-
- 112 propynoic acid, 1,1'-(1,2-ethanediyl) ester (Yne) and diazirine-grafted polycaprolactone tetrol
- 113 (CaproGlu; abbreviated to "Dz" for diazirine functional groups) are mixed into hybrid
- networks that result in both single-step activation with polychromatic light in the range of
- 115 320-500 nm (SH/Yne/Dz) and dual-step activation by monochromatic 405 nm light followed

by UVA activation (365 nm) for SH/Ene/Dz. The general hypothesis is that thiol/ene reaction 116 results in liquid-to-gel transition upon exposure to visible light (405 nm; Step 1) where 117 diazirines are stable. This 3-component hybrid gel can be formed in any desirable shape/size 118 prior to diazirine-to-carbene reaction, activated by UVA light (365 nm; Step 2;). The carbene 119 subsequently inserts into any type of solid surface resulting in covalent adhesion of 120 crosslinked hybrid network. The simple blending procedure of components with different 121 molecular geometries (Figure 2) provide a facile preparation method of injectable synthetic 122 123 biomaterials with a wide range of elastic moduli possible [19].



124

Figure 2. Chemical structures of individual components and nomenclature (in red font) used
for hybrid polymer network, crosslinked by light activation with the presence of TPO
initiator; CaproGlu formulation is polycaprolactone tetrol (PCLT; 1000 Da) grafted with 4-

- 128 [3-(Trifluoromethyl)-3H-diazirin-3-YL] benzoic acid (Dz-COOH; 230 Da) with
- 129 concentration of grafted Dz-COOH = 50% (2 diazirine functional groups per molecule).

- 131 Well-known thiol/ene and thiol/yne reactions are reported to occur by two mechanisms:
- thiol/ene radically mediated reactions and Michael-type additions [20, 21]. Thiol/ene free-

radical reactions are light-activated in the presence of photoinitiator with controlled

- 134 crosslinking kinetics [22]. This crosslinking reaction is particularly useful for acrylate
- systems in coatings industry, dental and tissue engineering applications [23, 24]. *In-situ* light
- activation with visible light or with UVA light (i.e. absorbed light energy dose: 10-20 J) is a
- 137 polymerization method of choice for synthetic implant design where the implant first takes
- shape/size of surgical site (Step 1) and subsequently adheres to solid substrate (Step 2;
- **Figure 1**). For the first time, this paper describes diazirine-based 2-step crosslinked hybrid
- polymer network with characteristics of double sided adhesive gel and controlled dynamicmechanical modulus.
- 142

143 2. EXPERIMENTAL SECTION

144 2.1 Materials

- 145 Polyethyleneglycol-diacrylate (Ene; **Figure 2**), diphenyl (2,4,6-trimethyl benzoyl) phosphine
- 146 oxide (TPO) and other reagents and solvents (KOH, KMnO₄, HCl, MgSO₄, 1,1-
- 147 carbonyldiimidazole (CDI), deuterated and pure dichloromethane (DCM), diethyl ether;
- 148 Et₂O) are purchased from Sigma (Singapore). Pentaerythritol tetrakis (3-mercaptopropionate;
- 149 SH; Figure 2), 4-[3-(Trifluoromethyl)-3H-diazirin-3-yl] benzyl alcohol (Dz-MeOH) and 2-
- 150 Propynoic acid, 1,1'-(1,2-ethanediyl) ester (Yne; **Figure 2**) are purchased from TCI
- 151 Chemicals (Japan). Polycaprolactone tetrol (PCLT; $M_w = 1000$ Da) is kindly donated by
- 152 Ingevity (CapaTM 4101; Lot No. HTX06P024).

2.2 Synthesis and preparation of 2-step light-activated hybrid polymer networks: thiol, acrylate and diazirine-grafted polycaprolactone tetrol

- 155 CaproGlu is diazirine-grafted PCLT (**Figure 2**) synthesized by previously published method
- 156 [25]. In brief, diazirine grafting is obtained by esterification reaction between PCLT and 4-[3-
- 157 (Trifluoromethyl)-3H-diazirin-3-YL] benzoic acid (Dz-COOH produced by Dz-MeOH
- 158 oxidation) conducted with 1,1- carbonyl diimidazole (CDI) used as a coupling agent. The
- molar ratio of Dz-COOH/PCLT = 2/1 is deliberately chosen to yield ~50% diazirine
- 160 conjugation. Hybrid polymer networks: Yne/Ene/SH/CaproGlu are prepared by mixing liquid
- 161 components (**Figure 2**) in predetermined concentrations into glass vials; 2-component
- 162 mixtures (SH/Yne and SH/Ene) and pure CaproGlu (Dz) are used as controls for crosslinking
- 163 method in photorheometry experiment (**Table 1**). The solvent (DCM) is added to the mixture

- 164 (approximately up to 10% w/v polymer/solvent) and vortexed 3 times (20 sec each time).
- 165 Stock solution of TPO (10% w/v; DCM) is added (maximum 0.1% w/w TPO/polymer) to the
- polymer solution and vortexed again 3 times (20 sec each time). The solvent is evaporated
- under vaccum to produce solvent-free hybrid polymer blends in the molar ratios displayed in
- **Table 1** (number of mmols of SH, Yne and Ene are normalized to CaproGlu set as 1 mmol).

Table 1. Molar ratios of hybrid networks with controls normalized to CaproGlu concentration (pure CaproGlu is used as Control-1 in all measurements) and light activation methods: 405 nm and 365 nm are two consequent activation steps used in dual curing photorheology experiment.

					Light activation		
Hybrid network composition	4-Thiol (SH)	2-Alkyne (Yne)	2-Alkene (Ene)	2-Diazirine (CaproGlu)	320-500 nm	405 nm	365 nm
CaproGlu (Control-1)	-	-	-	1	Х	\checkmark	\checkmark
SH/Yne (Control-2)	1	1	-	-	\checkmark	Х	Х
SH2/Yne2/Dz	2.3	2.3	-	1	\checkmark	Х	Х
SH/Ene (Control-3)	1	-	2	-	Х	\checkmark	\checkmark
SH2/Ene2/Dz	2.0	-	2.2	1	Х	\checkmark	\checkmark
SH1/Ene2/Dz	1.1	-	2.1	1	Х	\checkmark	\checkmark
SH1/Ene1.5/Dz	1.1	-	1.5	1	Х	\checkmark	

170

171 **2.3** Photorheometry analysis of hybrid polymer networks

Samples are analysed with Anton Paar MCR302 rheometer (SH/Yne/Dz: PP08 / 8 mm 172 diameter probe / 0.2 mm probe-base gap) and MCR302 rheometer (SH/Ene/Dz: PP10 / 10 173 mm diameter probe; 0.1 mm probe-base gap), equipped with UV-transparent glass base. The 174 following is an example of rheometry evaluation: rotational shear at 10 Hz for 30 seconds, 175 followed by dynamic shear for 120 seconds (1% amplitude; 10 Hz frequency) together with 176 photoirradiation (UV light; OmniCure Series 1500 UV Spot Curing System with 320-500 nm 177 bandpass filter (100 mW.cm⁻²) for 200 seconds (SH/Yne/Dz). Dual crosslinking (SH/Ene/Dz) 178 is performed with two different light sources: each 3-component network and controls (Table 179 180 1) are first subjected to rotational shear at 10 Hz for 60 seconds, followed by dynamic shear (1% amplitude; 10 Hz frequency) during which the samples are irradiated with 405 nm 181 exposure (power: 100 mW.cm⁻²; Thorlabs SOLIS-405C High-Power LED), followed by 365 182 nm UVA exposure (Convoy S2+ 365 nm Nichia UV Waterproof LED Flashlight; 100 183

mW.cm⁻²). Light diodes are calibrated with IL 1400 Radiometer. Storage (G') and loss (G'')
moduli are recorded over time during the dynamic shear. G' and G'' are measured both as a
function of irradiation time and energy (J.cm⁻²; referred to as "J" further in text).

187 2.4 Fourier-transform Infrared Spectroscopy (FTIR) analysis

FTIR spectra of 3-component hybrid networks (crosslinked by exposure to ambient light for 188 24 h) and pure CaproGlu are recorded before and after UVA activation (single components: 189 190 SH, Ene and CaproGlu are recorded as controls). Solid hybrid network sample is centred 191 between 2 UVA diodes (Supplementary Information; Figure S1-2) and activated from both sides (10 J each side). Liquid samples (pure components) are placed on glass slides and 192 193 activated with 10 J dose of UVA light. FTIR spectra are recorded before and after UVA activation in attenuated total reflection (ATR) mode at the following timepoints: before UVA 194 195 (neat), immediately after UVA activation and 10 min, 20 min, 30 min and 24 h post-UVA. FTIR spectroscopy experiment is performed using PerkinElmer Frontier IR equipped with 196 ATR sampling accessory. Spectra are recorded over accumulation of 8 scans at resolution 4 197 cm⁻¹, at range of 4000-600 cm⁻¹. The theoretical calculation of the concentration of diazirine 198 groups, contained at 1 cm² surface with 1 μ m thickness is performed by using the estimated 199 molecular weight of CaproGlu (1,500 Da) and the density of hybrid network determined 200 directly by weighing the samples with measured dimensions (Table S2). 201

202 2.5 Scanning electron microscopy analysis of crosslinked hybrid polymer network: 203 cross-section morphology profile

Crosslinked hybrid network is prepared by the same method as for FTIR spectroscopy and
lap shear adhesion experiments (10 J of absorbed light energy from each side of the square
sample (Supplementary Video_1) is cut in cross-sections and analysed with SEM. Samples
are subjected to platinum coating (90 s, chamber pressure <5 Pa at 20 mA). Images are
obtained by JSM 6360 SEM at an acceleration voltage of 5–20 kV and a working distance of
~15 mm.

210 **2.6** Adhesion strength analysis of hybrid networks on polymer surfaces

211 3-Component polymer networks in predetermined concentrations (**Table 1**) are casted into

212 petri dishes and irradiated with 405 nm diode for the total dose of 4 J. The samples are left to

- crosslink for 24 h under ambient conditions to produce ~1 mm thick films for peel test
- 214 (Figure S3). Samples are placed on collagen film fixed with cyanoacrylate onto glass slide.

The sample is placed onto a collagen surface with one collagen film on the top of the sample 215 thus forming collagen/HPN/collagen sandwich structure (Figure S4). Both glass slides with 216 fixed collagen film (bottom surface) and the collagen strips (top surface) are soaked in 217 purified water followed by removal of excess water with lint free paper. The sample is fixed 218 with an additional glass slide (from the top) with the aid of paper clips (Figure S4). The 219 sample is irradiated with UVA (365 nm) light from both sides with the total dose of 20 J (10 J 220 from each side). After UVA activation, the paper clips and the glass slide from the top are 221 222 removed. The sample is mounted onto a peel test cell and the top collagen film is pulled 223 upwards to record the peel strength (N/m) vs displacement (mm). Each HPN composition as well as CaproGlu control are measured with Series Force Measurement System (Chatillon 224 Force Measurement Products, USA) equipped with 100 N loading cell (n = 3). Gel-like 225 samples (representative SH1/Ene2/Dz; Supplementary Video_2) are cut in square films and 226 the weights / dimensions are recorded for each sample to estimate the density values of tested 227 materials. Sample cuts (~1 mm thickness) are produced with a surgical blade and are placed 228 on PMMA slide centered between 2 UVA diodes (top and bottom of the sample). A PET 229 230 sheet is placed on the top of the sample, hand-pressed with the aid of glass microscope slide, UVA diodes are turned ON, simultaneously to deliver the total dose of 20 J (counted together 231 from both sides; bottom through PMMA and top through glass + PET) to produce PMMA-232 hybrid network-PET sandwich structure for lap shear adhesion test. Shear adhesion strength 233 is measured with Series Force Measurement System (Chatillon Force Measurement Products, 234 USA) equipped with 100 N loading cell (n = 7). 235

236 2.7 Data processing

All the calculations and graphs are produced in OriginPro software.

238

239 **3. RESULTS**

240 **3.1** Hybrid networks with diazirine-grafted polymer crosslinker: the scope

241 The combination of polymers (Figure 2; Table 1) is hypothesized to form an instantaneous

hybrid gelation by: (i) single step curing with polychromatic light range of 320-500 nm

- 243 (thiol/alkyne/diazirine) or (ii) dual (2-step) crosslinking with monochromatic light
- wavelengths at 405 nm (thiol/alkene; Step 1) and UVA (diazirine-to-carbene; Step 2; Figure
- 1). The overall design intent gives a light activated hybrid network for use as double sided

tissue adhesive patches. CaproGlu's liquid polymer nature creates the opportunity to dissolve 246 free radical and step growth polymerization monomers to create stimuli-based biomaterials. 247 Light irradiation at specific wavelengths can be selectively chosen to activate the monomers. 248 The following thiol/ene and thiol/yne monomers are common hydrogel precursors: 4-249 functional thiol (SH), 2-functional alkene (Ene), 2-functional alkyne (Yne) and 2-functional 250 251 diazirine grafted polycaprolactone tetrol (CaproGlu; Dz). The exact mol ratios and respective nomenclature are shown in Table 1; for example, the network "SH2/Ene2/Dz" is a 3-252 component liquid mixture in the following molar ratio: PTHT/PEGDA/CaproGlu = 2/2.2/1253 254 (structures are shown in Figure 2). Neat CaproGlu and 2-component hybrids (SH/Yne and SH/Ene) are used as controls. Thiol/ene and thiol/yne crosslinked resins are selected for their 255 miscibility in CaproGlu and allow simultaneous or independent activation when doped with 256 visible light photoinitiators. Simultaneous activation is first evaluated with polychromatic 257 light in the range of 320-500 nm (2SH/2Yne/Dz; Table 1). Independent activation is 258 evaluated for the SH/Ene/CaproGlu hybrid elastomer by visible light (405 nm: thiol/ene 259 polymerization) [26], followed by CaproGlu activation with UVA light (365 nm: carbene-260

261 based crosslinking).

The specific monomers (SH, Yne and Ene; Figure 2) are chosen as they are capable of 262 bioresorption via ester hydrolysis similar to polycaprolactone-based materials. Surface-263 distributed diazirines are hypothesised to enable double sided adhesion by carbene insertion 264 upon UVA light activation. To investigate crosslinking kinetics, a custom photorheometer is 265 used to evaluate the real-time dynamic mechanical properties. One of the objectives is to find 266 compositions with low yield stress, also known as Bingham plastics (Table 1) that allow 267 substrate conformation for double sided adhesion. Structure-property relationships are 268 determined with respect to light energy exposure (J.cm⁻²; referred to as "J" further in text), 269 270 and tertiary ratios of the hybrid components. Qualitative analysis of surface chemistry is evaluated by ATR-FTIR spectroscopy to observe chemical reactions of surface-exposed 271 272 functional groups before and after light activation. All hybrid networks are tested for their peel strength when UVA-crosslinked between two hydrated collagen surfaces. Scanning 273 electron microscopy (SEM) evaluates the depth of CaproGlu activation through porous 274 morphologies and fundamental mechanical properties (lap shear adhesion strength and 275 toughness) are evaluated in regard to the cross-sectional depth of diazirine activation. 276

3.2 Dynamic mechanical properties and crosslinking kinetics profiles are tuned by the rule of mixtures between individual polymers

Thiol/yne/Caproglu hybrid (SH2/Yne2/Dz; the exact mol ratios are shown in Table 1) is 280 activated with polychromatic light in UVA/visible range (320-500 nm; mercury lamp) for 281 simultaneous activation of CaproGlu (UVA) and the thiol/yne network (visible light). Storage 282 (G') and loss (G'') moduli are recorded as a function of time and irradiation dose (Figure 283 **3A**). Gelation (G' = G'') of SH2/Yne2/Dz hybrid network is reached at the dose of 4.2 J 284 (absorbed per cm⁻² in all photorheometry experiments) which is almost three times higher 285 than gelation dose measured for pure CaproGlu (1.6 J; control-1) activated with mercury 286 lamp (320-500 nm; Supporting Information, Figure S5). The 2-component mixture SH/Yne 287 (control-2; Table 1) required a higher gelation dose of 6 J (Figure 3B). 288



289



This result demonstrates that introduction of CaproGlu into the SH/Yne/Dz mixture results in lowering the gelation point in simultaneous crosslinking process of thiol/yne and covalent carbene insertion. Upon absorption of light at the total dose of 20 J, the hybrid 3-component polymer network reaches dynamic modulus (G') value of 500 kPa (**Figure 3A**) that is an order of magnitude lower than value reached for 2-component control-2 (G' > 5 MPa; **Figure**

3B). Lower G' value likely results from emission of nitrogen upon diazirine photolysis and 302 formation of foam like structure [17, 18, 25]. Introduction of diazirine-grafted polymer into 303 thiol/yne reaction system results in 1-step reaction within seconds activated by polychromatic 304 mercury light (320-500 nm). This result expands the application of thiol/yne reaction that is 305 relevant in both bioorganic chemistry and polymer synthesis [27]. It should be noted that the 306 relatively high energy dose (20 J) is necessary to reach G' plateau for both SH2/Yne2/Dz 307 (Figure 3) and pure CaproGlu (Figure S5). This is a result of polychromatic light emitted 308 from mercury lamp (320-500 nm) where only the portion of the total absorbed light (320-390 309 310 nm) activates diazirine groups [17].

311 Thiol/ene precursors are selected for independent light activation investigations. These precursors are available with similar molar mass (489 Da and 566 Da; Figure 2) to closely 312 313 align polymer and functional group molar ratios. Independent 2-step light activation of three hybrid networks (Table 1) is evaluated by visible, followed by UVA activation. All 3-hybrid 314 315 network samples in **Table 1** with the two-step light activation are first sheared for 1 min followed by irradiation with 405 nm light (Step-1). After 2 min no further changes in 316 modulus are observed and the sample is irradiated using a UVA diode (365 nm) with a 6 J 317 dose (Step-2; Figure 4). Light activation intervals are marked with dashed lines in Figure 4 318 and Figure S6. The representative evolution of G' for SH1/Ene2/Dz hybrid is compared to 319 pure CaproGlu (control-1) and 2-component diazirine-free SH/Ene mixture (control-3) as 320 shown in Figure 4A. Both SH/Ene (control-3) and SH1/Ene2/Dz hybrid show an increase in 321 G' values upon activation at 405 nm, indicating free-radical (thiol/ene) polymerization in the 322 presence of CaproGlu that remains unreactive until the UVA light is turned on. 323

According to rheometry data in **Figure 4A**, the rapid free-radical polymerization appears to

be faster in comparison to carbene-induced CaproGlu crosslinking (judging from the G' vs

irradiation time required to reach G' plateau). The 2-component control-3 (SH/Ene; no

327 CaproGlu) shows no change in G' upon UVA activation. Neat CaproGlu indicates liquid-to-

biorubber transition when activated either by polychromatic light (Figure S5) or

329 monochromatic UVA (365 nm) diode (Figure 4B). However, 3-component hybrid networks

- (Table 1) undergo 2-step crosslinking without indication of gelation point (Figure 4C;
- **Figure S6**). Note that pure CaproGlu takes 6 J UVA dose to reach G' value of ~170 kPa
- (Figure 4B) consistent with previously published work on CaproGlu bioadhesive [17, 25].



Figure 4. 2-step energy directed crosslinking of liquid-to-solid hybrid network (3-component 334 compositions are specified in Table 1) analysed with photorheometry by using different 335 wavelengths: (A) representative dynamic change of storage moduli (G') of 3-component 336 thiol/ene/CaproGlu hybrid network (SH1/Ene2/Dz) upon crosslinking at 405 nm (Step 1) 337 followed by activation at 365 nm (Step 2) in comparison to 2-component thiol/ene (SH/Ene; 338 control-3) and pure CaproGlu (control-1); the power of both diodes is adjusted to the total of 339 10 J.cm^{-2} dose (diode power = 100 mW.cm^{-2}); (inset) SH1/Ene2/Dz transitioned into a 340 flexible film after dual irradiation; (B) dynamic change of G' and loss modulus (G'') upon 341 light irradiation recorded for pure CaproGlu with indicated gelation point (G' = G''); (C) G' 342 343 and G" change with irradiation energy recorded for SH1/Ene2/Dz; (D) normal force caused 344 by the volume expansion of polymer networks upon light-activated crosslinking recorded for 3-component hybrid networks compared to 2-component (SH/Ene; control-3) and pure 345 346 CaproGlu (control-1). All data points are measured over time (bottom) and energy dose $(J.cm^{-2}; top)$ 347

348 Unlike controls that show no indication of crosslinking by either visible light (405 nm; pure

CaproGlu) or by UVA light (365 nm; 2-component SH/Ene), 3-component hybrid network

results in 2-step crosslinking process that is first activated by visible light (thiol/ene) followed

by UVA activation of diazirine to reach the maximum at 920 kPa recorded for SH1/Ene2/Dz

352 hybrid network (Figure 4A and C). Hybrid elastomeric films could be peeled off the

rheometer probe after a dual curing experiment as shown in Fig. 4A-inset (the rheometry

354 results are summarised in **Table S1**).

355 The photorheometer simultaneously assesses volumetric shrinkage by applying normal force 356 to maintain a predetermined sample thickness. The normal force to the rheometer probe is 357 monitored for both hybrid networks and controls 1-2 (pure CaproGlu and 2-component, diazirine-free SH/Ene mixture, respectively) and results are shown in Figure 4D. Control-2 358 demonstrates a decrease of normal force with exposure to 100 mW.cm⁻² power 405 nm diode 359 within seconds. The SH/Ene molar ratio in control-2 (**Table 1**) is 1/2 for the highest density 360 361 of crosslinking between 4-functional thiol and 2-functional acrylate. A drop in normal force is evidence for shrinkage and an increase signifies matrix expansion. Volume shrinking is 362 observed under visible light activation due to Michael addition and free radical crosslinking. 363 3-Component hybrid networks (Table 1) all resulted in sample volume shrinkage upon 405 364 nm activation (Step 1) as evident from drop in normal force (see Figure S7). CaproGlu 365 shows no change in normal force during Step 1 light activation (405 nm; 4 J; Figure S7) 366 suggesting the inert nature of diazirine groups towards visible light (405 nm). Under UVA 367 light exposure, diazirine photolysis releases molecular nitrogen as a byproduct that in turn 368 causes foaming [17, 25]. Volume expansion is evident in both pure CaproGlu (control-1) and 369 3-component hybrid samples (Figure 4D). This further supports the hypothesis of hybrid 370 network crosslinking with independent light wavelengths. 371

The hybrid networks are observed to spontaneously solidify over 24 h (no UVA activation)

373 when exposed to laboratory ambient environment, which may be due to visible light

activation (Figure S2). Note that SH1/E1.5/Dz hybrid network (refer to Table 1 for exact

- mol ratio) results in viscoelastic solids (i.e. Bingham plastic) material (Figure S1). The
- elastomeric hybrid composition can be cut into solid, double sided adhesive polymer
- 377 (network SH1Ene2/Dz; Table 1; Supplementary Video_1) and forms a solid material after
- 378 24 h exposure to ambient light; this formulation has a $\sim 1/1$ ratio thiol/ene (**Table 1**). The
- absence of gelation points in hybrid networks (Figure 4C; Figure S6) is possibly a
- 380 consequence of immediate reaction upon mixing under ambient light condition as the

photorheometry analysis is performed within 30 min after sample preparation. The absence of 381 gelation points (recorded 30 min after mixing) are consistent with the macroscopic 382 appearance of hybrid networks that form either solid gels or Bingham plastic materials even 383 without direct irradiation by 405 nm light diode (Figure S1). Note that stoichiometric 384 SH/Ene ratio in SH1/Ene2/Dz (4-arm thiol and 2-arm acrylate; Figure 2) results in highest G' 385 = 28 kPa upon 405 nm activation in comparison to SH2/Ene2/Dz (16 kPa) and 386 SH1/Ene1.5/Dz (7 kPa) where unreacted thiol acts as plasticizer (Figure 4; Figure S6; Table 387 S1). Although the highest relative concentration of diazirines is in SH1/Ene1.5/Dz, the 388 389 stoichiometric thiol/acrylate ratio (SH1/Ene2/Dz) results in the highest recorded G' after second activation step with UVA (920 kPa) compared to 410 kPa and 630 kPa recorded for 390 SH2/Ene2/Dz and SH1/Ene1.5/Dz respectively (Figure 4; Figure S6). Carbene reacts both at 391 the surface and within the bulk of materials, depth limited by macromolecule scattering. The 392 unreacted thiol within the hybrid networks might act as carbene scavenger thus reducing the 393 extent of surface reaction with both base and the probe of photorheometer, that in turn results 394 395 in relatively low modulus.

396 3.3 FTIR spectroscopy identifies covalent crosslinks within hybrid networks

FTIR spectroscopic analysis is performed to qualitatively observe the depletion of reactive 397 functional groups upon photoreactions (Figure 5), namely: out of plane –C=C– stretch 398 vibrations at 810 cm⁻¹ (2-Ene) [21, 28], S-H absorption peaks at 2570 cm⁻¹ (4-SH) [29], 399 diazirine ring at 1634 cm⁻¹ [25, 30] and diazoalkane intermediate peaks at 2092 cm⁻¹ [25, 31]. 400 The network SH1/Ene2/Dz (the exact molar ratio is shown in Table 1) is chosen for this 401 experiment to prove the hypothesis that under visible light only thiol reacts to completion 402 with acrylate while diazirine groups remain unreacted. Figure 5A shows full spectra of both 403 hybrid SH1/Ene2/Dz and neat components (no UVA activation) while Figure 5B is a 404 magnified region used in this analysis. Note that the peaks at 1640-1610 cm⁻¹ of Ene 405 (assigned to -C=C- from acrylate) overlap with diazirine (-N=N-; 1630 cm⁻¹) and -C-N 406 (1610 cm⁻¹) [25] peaks from CaproGlu and therefore could not be used to observe reaction of 407 acrylates [32]. For that reason, the peak at 810 cm⁻¹ is selected to analyse Ene (acrylate) 408 reaction within hybrid matrix (Figure 5B-inset). Both thiol (2570 cm⁻¹) and acrylate (810 cm⁻¹) 409 ¹) disappear in SH1/Ene2/Dz sample after photocuring by ambient light with estimated 410 411 degree of Ene conversion of ~80% even before activation with UVA light (Figure S8).



412

Figure 5. FTIR spectral regions recorded for representative 3-component thiol/ene/CaproGlu 413 hybrid network (SH1/Ene2/Dz) and pure polymer components (structures and nomenclature 414 of individual component are shown in Figure 1): (A) full FTIR spectral region (no UVA); 415 (B) diazirine (1631 cm⁻¹; CaproGlu and SH1/Ene2/Dz), -C-N (1610 cm⁻¹), -S-H (2570cm⁻¹; 416 pure 4-functional thiol: SH) peaks; (inset) -C=C- stretch vibration at 809 cm⁻¹ from 417 diacrylate groups (Ene; no UVA); (C) disappearance of diazirine peak (1634 cm⁻¹; dashed 418 line) upon UVA activation; (D) diazoalkane (2094 cm⁻¹) persistence over the period of 24 h 419 upon UVA activation of 3-component hybrid network (SH1/Ene2/Dz) and pure CaproGlu. 420

422 Upon UVA activation, diazirine undergoes photolysis as recorded from disappearance of

423 -N=N- peak at 1630 cm⁻¹ (**Figure 5C**). Two products are possible from diazirine photolysis:

- 424 carbene and diazoalkane. The semi-stable nature of the diazoalkane is demonstrated by the
- 425 presence of 2090 cm⁻¹ peak absorbance of both SH1/Ene2/Dz hybrid and neat CaproGlu

(control) 30 min post-UVA activation (Figure 5D). Previously published FTIR results 426 provide the evidence that the diazoalkane decay kinetics is dependent on the functional 427 groups adjacent to diazirine [33]. The intermolecular environment (in this case SH/Ene 428 mixture) is likely to influence the fate of diazoalkane, however this requires dedicated 429 kinetics study in future research. It is hypothesized that the adhesion to solid surfaces of 3-430 component hybrid network is facilitated by the fraction of the diazirine groups that are 431 distributed at the hybrid surface after the first crosslinking step. From theoretical estimation 432 (according to molar ratio in Table 1) the SH1/Ene2/Dz network contains 70 nmol.cm⁻² of 433 diazirine groups for the sample thickness of 1 µm (the density of this particular network is 1.1 434 g.cm⁻³; **Table S2**). Indeed, diazirine groups are detected in ATR-FTIR experiment (Figure 435 **5C**) with penetration depth of ATR probe to be ~ 100 nm [34]. With evidence that diazirine 436 surface groups remain unreacted, double sided adhesives are possible. 437

3.4 UVA activation of diazirine component results in porous surface micro-morphology detected by SEM

440 As indicated by the double sided adhesive nature of the 3-component SH1/Ene2/Dz hybrid network (Table 1), UVA irradiation activates the sample surface causing covalent insertion 441 of carbenes onto solid polymer interfaces (Supplementary Videos 1 and 2). SEM analysis of 442 cross-section (Figure 6) is performed to examine the penetration depth of diazirine activation 443 visually observed by the colour change (from white opaque to yellow; Figure 4A-inset). 444 Arrows pointing out from Figure 6A-inset indicate different parts of SH1/Ene2/Dz cross-445 section (surface Figure 6B and bulk Figure 6C) analysed with SEM. UVA activation of 446 CaproGlu causes evolution of molecular nitrogen that in turn results in porous crosslinked 447 448 matrix [25]. From Figure 6A-B, the porous matrix is formed at the surface of the hybrid network with estimated penetration depth of molecular nitrogen in the range of 100-150 µm 449 450 while the middle portion of the sample (**Figure 6C**) shows micro-wrinkled morphology, characteristic for elastomeric surfaces [35]. The pore size generated by molecular nitrogen 451 (diazirine photolysis) is in the range of 5-10 µm (Figure 6B), 10x smaller than neat 452 CaproGlu with pore size of 50-100 µm (Figure 6D). 453



454

Figure 6. SEM images of cross-section of 3-component thiol/ene/CaproGlu hybrid polymer
network after 10 J.cm⁻² of UVA activation indicating the outer porous and inner
homogeneous morphology of UVA activated sample - 2 parts of the sample, surface and bulk
(inset photography) are indicated by arrows: (A-C) SH1/Ene2/Dz; (D) pure CaproGlu
(control-1) activated with 10 J.cm⁻² of UVA (magnification bar: 100 µm).

Decrease of pore size is a direct consequence of material properties – unlike the crosslinked 460 hybrid network that results in a semi-solid material, liquid (neat) CaproGlu results in lower 461 stress that inhibits nitrogen bubble expansion. Knowing the penetration depth of CaproGlu 462 crosslinking within the SH1/Ene2/Dz hybrid network (100-150 µm; Figure 6A) attention 463 should be given to the rheometry base-gap probe that is set for 100 µm, and therefore 464 dynamic moduli (G') in Figure 4 and Figure S6 are representative of complete diazirine 465 (CaproGlu) crosslinking within the total volume of analysed hybrid network sample (Figure 466 4A-inset) as confirmed by SEM. Elastomeric wrinkled morphology [36] in the bulk of 467 SH1/Ene2/Dz hybrid is in line with visual observation of the material (Supplementary 468 Video 1). 469

470 Crosslinked SH1/Ene2/Dz (Step 1; 405 nm) does not allow pore expansion deeper and larger

- than measured by SEM, however, future work could demonstrate that the pore size might be
- 472 controlled by the following parameters: (1) concentration of grafted diazirine and molecular
- 473 weight of polycaprolactone polyol; (2) CaproGlu concentration within composite (hybrid)
- 474 network; and (3) crosslinking density activated with visible light, prior to UVA activation.
- Furthermore, diazirine can be activated with longer wavelength at 445 nm (with the aid of
- 476 photocatalyst) [18] that will possibly allow light-activated crosslinking deeper than $150 \,\mu m$.

477 **3.5 Double sided adhesive hybrid network properties**

Hybrid network SH1/Ene2/Dz (Table 1) has demonstrated light sensitivity in both visible 478 479 light (405 nm; G' = 28 kPa at 4 J dose) and UVA (365 nm; G' = 920 kPa at 6 J dose) resulting in the highest dynamic moduli in comparison to other hybrid networks (Figure 4; 480 481 Table S1). Predetermined SH/Ene ratio (1/2) results in complete thiol/ene reaction (as evident from FTIR spectroscopic analysis; Figure 5B) forming a gel polymer network after 482 activation by visible light at low energy dose (4 J; Figure 4A). Rheometry results 483 484 demonstrate that CaproGlu was unaffected by visible light and could be activated independently – visible light activation of SH1/Ene2/Dz hybrid allows mm thick specimens 485 to be setup even in the presence of CaproGlu (Supplementary Videos 1 and 2). Diazirines 486 are known to convert into carbene upon UVA activation that in turn results in unselective 487 crosslinking, both polymer bulk (intermolecular) and at any proximate surface (i.e. polymer, 488 tissue proteins) [17]. Peel strength experiment is performed for hybrid networks where the 489 samples (Figure S3) are placed between two hydrated collagen surfaces (Figure S4) and 490 subsequently activated with UVA light from both sides of the "sandwich" structure with the 491 492 total dose of 20 J (10 J delivered through each collagen film). The obtained results are compared to the following controls: 1) neat CaproGlu before and after UVA activation; 2) 493 494 two-component blend SH/Ene (without CaproGlu) previously crosslinked under ambient light (Figure S3) and subsequently irradiated with 20 J of UVA light through collagen sheets; 495 496 3) hybrid polymer networks fixed between two collagen films without UVA activation (Figure 7). 497

498



500

501 Figure 7. Peel adhesion strength measured for HPN samples (and pure CaproGlu used as control) activated through hydrated collagen surfaces with 2 UVA diodes - each side of all 502 samples absorbed the UVA energy = 10 J.cm^{-2} (total absorbed dose = 20 J.cm^{-2}): (A) peel 503 strength measured for HPN samples compared to CaproGlu control (n = 3; ANOVA: p<0.05; 504 SD = statistically different); (B and C) representative peel strength profiles collected for pure 505 CaproGlu and SH1/Ene2/Dz respectively (data between 2 vertical lines is used to calculate 506 average peel strength for each sample -2 mm after the beginning of the test and 2 mm before 507 508 adhesion failure); (D) representative photographs of collagen strips with crosslinked CaproGlu (left) and SH1/Ene2/Dz (right) samples after peel strength experiment. 509

All tested hybrid networks resulted in peel adhesion strength (after UVA activation) of 8 ± 3 N/m with a relatively large standard deviation and without significant difference between samples (**Figure 7A**). All the raw data used to calculate values in **Figure 7A** are shown in **Figures S9-11**. This result corresponds with photorheological properties where no change in dynamic modulus could be recorded upon UVA activation without the presence of diazirine

component (Figure 4). Hybrid polymer networks SH1/Ene2/Dz and SH1/Ene1.5/Dz show 516 adhesion when pressed between hydrated collagen sheets with the aid of paper clips (Figures 517 S4 and S11). However, the values of 5 ± 2 N/m and 1.2 ± 0.7 N/m (SH1/Ene2/Dz and 518 SH1/Ene1.5/Dz respectively) are significantly lower than peel adhesion strength recorded 519 after UVA activation (Figure 7A). Both neat CaproGlu and two-component network 520 521 (SH2/Ene2) controls do not show any adhesion that could be recorded without UVA irradiation. Together with FTIR analysis (Figure 5), this result supports the hypothesis that 522 523 the diazirine groups are present at the hybrid network surface and result in adhesion when 524 activated with UVA light at the hydrated biopolymer interface (collagen). Neat CaproGlu results in significantly higher peel strength after UVA activation (19 ± 9 kPa) when compared 525 to tested hybrid networks (Figure 7A). This result is expected because diazirine groups in gel 526 networks (Figure S3) are diluted and thus cause lower peel strength than crosslinked 527 CaproGlu. Furthermore, during the peel experiment, the fracture of crosslinked polymer 528 529 propagates evenly for CaproGlu as the fracture strength is distributed over the entire sample volume (Figure 7B; Figure S9). Representative peel adhesion data for hybrid polymer 530 531 network (SH1/Ene2/Dz; Figure 7C) shows peak and trough indicating larger strain fracture toughness. CaproGlu fails cohesively during peel test [17, 18], unlike in crosslinked hybrid 532 533 networks that demonstrate adhesive failure due to diazirine dilution (representative photographs are shown in **Figure 7D**). Increased toughness by formation of a hybrid network 534 matrix correlates with rheology results (Figure 4) that indicate crosslinked matrix upon 535 thiol/ene reaction activated by visible light (Step-1; Figure1). 536

Cohesive nature of CaproGlu adhesive is consistent with previously published results on pure 537 CaproGlu used in artery anastomosis aided with polycaprolactone (PCL) mesh tape [17]. The 538 biorubber nature of hybrid network (Supplementary Video_2) allows compliance with soft 539 540 tissues and potentially could prevent implant failure and injuries due to biomechanical mismatch. The elastic nature of crosslinked hybrid network is further investigated in lap shear 541 542 adhesion test to polymer surfaces that resulted in ultimate adhesion strength of 160 ± 50 kPa (representative SH1/Ene2/Dz; Figure S12A-B). The elastic behaviour is also evident from 543 modulus vs strain diagram (Figure S12C) calculated as the first derivative of stress vs strain 544 function. The modulus value increases from 140 kPa to the maximum value of 197 kPa at 545 0.42 (mm/mm) strain and sustains strain > 0.8 (marked with dashed arrow in **Figure S12C**). 546 The strain energy density (toughness) of the hybrid network is measured to be 80 ± 40 kJ.m⁻³ 547 548 (Figure S12D) for the samples of ~2 mm thickness (Figure S12B-inset; Table S2). The

toughness of the hybrid network is an order of magnitude lower that the toughness value 549 reported for thin CaproGlu film (~20 µm) activated at biological surfaces (~20 MJ.m⁻³ 550 collagen/CaproGlu/porcine skin) [18]. This result indicates that UVA activation is spatially 551 limited by UVA penetration depth through hybrid polymer network since all measured 552 samples failed cohesively (Figure S13C). The measured lap shear adhesion strength of 553 hybrid network is comparable to pure CaproGlu (170 ± 10 kPa; Figure S13). Unlike the peel 554 test performed with irregular, hydrated collagen films that results in adhesive failure (Figure 555 7), the hybrid network failed cohesively when crosslinked on flat, fully transparent polymer 556 557 surfaces (PET and PMMA; Figures S12 and S13). The diazirine adhesion is stronger than thiol/ene crosslinked network as evident by repetitive cohesive failure in lap shear adhesion 558 tests. Due to the cohesive nature of the double sided adhesive hybrid network, the diazirine 559 adhesion strength remains unknown. It is evident that the diazirine surface adhesion is 560 stronger than thiol/ene cohesive strength that always fails first. 561

562

563 **4. DISCUSSION**

Hybrid polymer blends that incorporate independent light activated crosslinking mechanisms 564 have been explored for the first time. Hybrid composites of thiol/ene and carbene-based 565 biomaterials serve as a model system to explore how viscous formulations can be rapidly 566 photocured into tough bioelastomers. The design requires liquid precursors that are miscible 567 568 yet remain relatively inactive. Liquid mixtures of CaproGlu, ester alkyne, PEG-based alkene, 569 and multi-arm thiol were miscible and chemically inert (Figure 2). Both alkenes and alkynes react with thiols via radical polymerization [37] and require a visible light-activated initiator 570 571 - for independent activation between the two polymer networks [26]. CaproGlu is inert to free-radical or visible light exposure, a considerable advantage for gamma sterilization and 572 573 ambient shelf stability [17]. Upon photoactivation, CaproGlu emits nitrogen, resulting in a 574 porous biorubber in both neat CaproGlu and hybrid polymer networks presented in this work. 575 The surface porosity of implanted biomaterials is known to accelerate tissue resorption while the tunable elastic modulus of hybrid polymer network is possible in the range of 400-950 576 577 kPa in broad wavelength light activation (320-500 nm) and focused activation by visible light (405 nm) followed by crosslinking and surface adhesion activated by UVA (365 nm) light. 578 Diacrylate/tetrathiol system has also shown crosslinking under ambient light. However, the 2-579

step crosslinking might be possible with other multifunctional alkenes to produce systemswith higher stability and more controlled photocuring mechanism [38].

582 The solvent-free hybrid network material allows for a flexible liquid-to-elastomer transition through various optical stimulation profiles. Thiol/ene crosslinking proceeds in the presence 583 of inert and transparent CaproGlu, resulting in dynamic modulus transition from liquid to ~30 584 kPa (modulus) gel upon activation with visible light (405 nm). Subsequent exposure to UVA 585 (365 nm), activates non-specific carbene insertion. The hybrid networks display a relatively 586 587 broad range of shear modulus from 10-800 kPa, which can be easily tuned through both optical exposure and precursor molar ratio. The modulus near the surface (100 µm depth) of 588 589 the hybrid network may allow modulus gradients to obtain matching elasticity profiles [39]. Apart from the potential control over depth of CaproGlu crosslinking, the porous surface of 590 591 hybrid network (pore size: $\sim 10 \,\mu$ m) is beneficial for tissue engineering strategies where neovascularization/cell migration is facilitated through interfacial porous structure of 592 593 implanted scaffolds [40].

594 Due to the homogeneous mixing of all three components, diazirine groups are present at the surface of the hybrid network. Hybrid polymer networks result in adhesion to both dry 595 substrates and wet collagen surfaces. Surface diazirine groups (from CaproGlu component) 596 resulted in carbene covalent insertion onto solid polymers (PMMA and PET). This covalent 597 598 insertion adheres hybrid network to polymers by reaching ~160 kPa of adhesion strength, limited by cohesive failure after applied lap shear adhesion stress. The penetration depth of 599 600 diazirine photolysis was found to be in the range of 100-150 µm while the diazirines 601 embedded within the network presumably remain unreacted. Diazirines are known to degrade 602 into ketones, alcohols, ethers or other chemical groups [41]. Both kinetics of diazirine degradation (photolysis) and the nature of degradation products are dependent on the 603 604 chemical environment [33, 41]. However, the results in this paper indicate possibility of double adhesive tape with 100 µm thickness where all diazirine groups are reacted, both 605 606 within the bulk of material and at the substrate interface.

607 Low molar mass diazomethane precursors require precautions due to their explosive nature

[42]. However, trifluoromethyl diazirine-based compounds (that lead to carbene and

609 diazoalkane intermediates) are known for their stability with reported crosslinking activation

that initiates at 110 °C [43]. CaproGlu synthesis reaction is stable (no detectable exothermic

effect) at 40 °C with exceptional shelf stability even after 25 kGy gamma sterilization, ergo

the aryl-diazirine is inert to free-radical exposure and most nucleophilic functional groups—a 612 claim few crosslinking groups hold [17, 25]. Apart from the stability of the diazirine used for 613 polymer grafting, polycaprolactone tetrol (PCLT) precursors are available in food-grade 614 quality and therefore present few risks towards medical devices. PCLT belongs to the 615 platform of PCL-based biodegradable materials that are known to undergo ester hydrolysis 616 617 and physiological elimination of degradation products goes through well-defined metabolic reactions such as citric acid and fatty acid pathways [69]. Polymerization of 618 thiol/ene/CaproGlu hybrid networks, initiated by exposure to gamma irradiation, may be 619 620 exploited for selective activation or depletion of acrylates without need of photoinitiators. CaproGlu would remain intact and to provide the same on-demand crosslinking / adhesion 621 characteristics. In addition, CaproGlu can be activated with visible light (445 nm) when 622 mixed with photocatalysts, which opens up possibilities for dual crosslinking activated by 623 two distinct visible wavelengths [18]. Unlike NHS-grafted bioadhesives [13], grafting of 624 625 carbene-generating diazirine onto liquid polycaprolactone polyols resulted in bioadhesive with non-discriminated covalent insertion to both hydrated biologically-derived surfaces and 626 627 solid synthetic polymers [17]. CaproGlu is one example of the emerging carbene-based bioadhesive platform. The liquid polymer requires no refrigeration or rehydration and can be 628 629 processed into ready-to-use implantable medical devices that are stable to gamma sterilization, a key attribute for industrial scale-up. CaproGlu has displayed little to no 630 inflammation tested in vivo [17] and low-risk skin sensitization in vitro (OECD-regulated 631 genotoxicity and sensitization tests) [25]. 632

Thiol/ene crosslinking is available for many different polymer systems, including acrylate-633 grafted polysaccharides and PEG macromolecules with a wide range of molecular weights 634 and geometries [44]. In particular, PEG diacrylate is known commercial photopolymer 635 636 precursor, available in a wide range of molecular weights [45]. PEG-based polymer networks are known for their application in hybrid bioprinting technology [46] and the ester bonds aid 637 638 miscibility within CaproGlu, allowing solvent-free mixing. In addition, polycaprolactone polyols (triols and tetrols) are readily available with molecular weights between 300 and 2000 Da, 639 allowing a library of materials with various viscoelastic properties [47]. Hybrid systems 640 mentioned herein could be extended beyond hydrophobic PCLTs. Amphiphilic formulations 641 could be designed with dendrimers [48-50]. This extends diazirine activation method to 642 applied voltage (Voltaglue) providing that the crosslinkers are dispersed in conductive 643 644 medium. The choice of initiator would determine the wavelength of light used for free-radical

- activation. For example, UV-active Igracure 2929 can be replaced with Eosin Y activated
- 646 with visible light (405 nm) to crosslink acrylate-thiol systems [51]. However, it is also known
- 647 that photoinitiators pose risks as toxic leachates [52]. Hybrid networks based on
- thiol/ene/carbene may eliminate photoinitiators through gamma initiation. Future work will
- 649 explore this process to form sterile double sided adhesives.
- 650

651 5. CONCLUSION

Diazirine-grafted polycaprolactone (CaproGlu) can be mixed with acrylates, thiols and 652 alkynes to form hybrid polymer networks with a high degree of control over material 653 properties. The dual curing macromolecular systems are independently activated by visible 654 light followed by mild UVA activation. When activated by visible light, polymer hybrid only 655 partially crosslinks into gel-like material and remains reactive for subsequent adhesion onto 656 solid surfaces by on-demand UVA activation of surface diazirine groups. Unselective 657 crosslinking of diazirine-generated carbene enables chemical anchoring of double sided 658 adhesive gels to any types of solid substrates without surface pre-treatment that is normally 659 required for formation of interfacial heterogenous chemical bonding. These attributes of 660 CaproGlu crosslinking formulation demonstrate multifunctional nature, both intermolecular 661 and interfacial crosslinking, that would lead towards biomedical applications with careful 662 selection of network components without need of solvents or photoinitiators. 663

664 CRediT authorship contribution statement

- 665 Ivan Djordjevic: Investigation, Formal analysis, Data curation, Writing original draft.
- 666 Gautama Wicaksono: Data curation, Methodology. Manisha Singh: Data curation,
- 667 Methodology. Elizabeth G. Ellis: Data curation, Methodology. Maher A. Alraddadi: Data
- 668 curation, Methodology. Andrew P. Dove: Conceptualization, Supervision, Writing review
- 669 & editing. Terry W.J. Steele: Conceptualization, Formal analysis, Supervision, Writing –
- 670 review & editing, Funding acquisition.

671 Declaration of Competing Interest

- T.W.J. Steele and I. Djordjevic are co-inventors of the following IP: Hygroscopic,
- 673 Crosslinking Coatings and Bioadhesives; PCT/SG2018/050452. Authors declare no
- 674 competing interests. CaproGlu is an abbreviation for this technology and is not trade marked.

676 Acknowledgements

- 677 This work is funded by Institute of Advanced Studies Birmingham Visiting Fellowship,
- 678 'Strong Elastomeric Bioadhesives Towards Tendon Repair'; Ministry of Education Tier 1
- 679 Grant RT07/20: Fiber-optic orthopaedic implants for bone-implant adhesion, Ministry of
- Education Tier 2 Grant (MOE2018-T2-2-114): CaproGlu, Double sided wet-tissue adhesives,
- 681 NTUitive POC (Gap) Fund NGF/2018/05: Aesthetic Applications of CaproGlu Bioadhesives,
- and A*STAR IAF PP Grant (H19/01/a0/0II9): CathoGlu Bioadhesives-preventing catheter
- 683 extravasation and skin infections.

684

685 **References:**

- 686 [1] S. Czarnecki, T. Rossow, S. Seiffert, Hybrid Polymer-Network Hydrogels with Tunable
- 687 Mechanical Response, Polymers 8(3) (2016) 82.
- [2] M.A. Haque, T. Kurokawa, J.P. Gong, Super tough double network hydrogels and their
- application as biomaterials, Polymer 53(9) (2012) 1805-1822.
- [3] J.P. Gong, Why are double network hydrogels so tough?, Soft Matter 6(12) (2010) 2583-2590.
- [4] H. Yuk, T. Zhang, S. Lin, G.A. Parada, X. Zhao, Tough bonding of hydrogels to diverse
 non-porous surfaces, Nature materials 15(2) (2016) 190-196.
- [5] J. Deng, Z. Dai, J. Yan, M. Sandru, E. Sandru, R.J. Spontak, L. Deng, Facile and solvent-
- free fabrication of PEG-based membranes with interpenetrating networks for CO2 separation,Journal of Membrane Science 570-571 (2019) 455-463.
- [6] M. Sangermano, W. Carbonaro, G. Malucelli, A. Priola, UV-Cured Interpenetrating
- Acrylic-Epoxy Polymer Networks: Preparation and Characterization, Macromolecular
 Materials and Engineering 293(6) (2008) 515-520.
- 700 [7] V. Granskog, O.C.J. Andrén, Y. Cai, M. González-Granillo, L. Felländer-Tsai, H. von
- 701Holst, L.-A. Haldosen, M. Malkoch, Linear Dendritic Block Copolymers as Promising
- 702 Biomaterials for the Manufacturing of Soft Tissue Adhesive Patches Using Visible Light
- Initiated Thiol–Ene Coupling Chemistry, Advanced Functional Materials 25(42) (2015)
 6596-6605.
- [8] A.H.C. Anthis, X. Hu, M.T. Matter, A.L. Neuer, K. Wei, A.A. Schlegel, F.H.L. Starsich,
- 706 I.K. Herrmann, Chemically Stable, Strongly Adhesive Sealant Patch for Intestinal
- Anastomotic Leakage Prevention, Advanced Functional Materials 31(16) (2021) 2007099.
- 708 [9] L. Zeng, J. He, Y. Cao, J. Wang, Z. Qiao, X. Jiang, L. Hou, J. Zhang, Tissue-adhesive and
- highly mechanical double-network hydrogel for cryopreservation and sustained release of
 anti-cancer drugs, Smart Materials in Medicine 2 (2021) 229-236.
- 711 [10] H.Y. Jung, P. Le Thi, K.-H. HwangBo, J.W. Bae, K.D. Park, Tunable and high tissue
- adhesive properties of injectable chitosan based hydrogels through polymer architecture
 modulation, Carbohydrate Polymers 261 (2021) 117810.
- 714 [11] S. Nam, D. Mooney, Polymeric Tissue Adhesives, Chemical Reviews (2021).
- 715 [12] X. Chen, H. Yuk, J. Wu, C.S. Nabzdyk, X. Zhao, Instant tough bioadhesive with
- triggerable benign detachment, Proceedings of the National Academy of Sciences 117(27)
- 717 (2020) 15497-15503.

- 718 [13] H. Yuk, C.E. Varela, C.S. Nabzdyk, X. Mao, R.F. Padera, E.T. Roche, X. Zhao, Dry
- double-sided tape for adhesion of wet tissues and devices, Nature 575(7781) (2019) 169-174.
- [14] M.H. Turabee, T. Thambi, D.S. Lee, Development of an Injectable Tissue Adhesive
- Hybrid Hydrogel for Growth Factor-Free Tissue Integration in Advanced Wound
- Regeneration, ACS Applied Bio Materials 2(6) (2019) 2500-2510.
- [15] A. Shagan, W. Zhang, M. Mehta, S. Levi, D.S. Kohane, B. Mizrahi, Hot Glue Gun
- Releasing Biocompatible Tissue Adhesive, Advanced Functional Materials 30(18) (2020)1900998.
- 726 [16] N. Oliva, S. Shitreet, E. Abraham, B. Stanley, E.R. Edelman, N. Artzi, Natural Tissue
- Microenvironmental Conditions Modulate Adhesive Material Performance, Langmuir 28(43)
 (2012) 15402-15409.
- [17] I. Djordjevic, O. Pokholenko, A.H. Shah, G. Wicaksono, L. Blancafort, J.V. Hanna, S.J.
- Page, H.S. Nanda, C.B. Ong, S.R. Chung, A.Y.H. Chin, D. McGrouther, M.M. Choudhury,
- F. Li, J.S. Teo, L.S. Lee, T.W.J. Steele, CaproGlu: Multifunctional tissue adhesive platform,
 Biomaterials 260 (2020) 120215.
- [18] I. Djordjevic, G. Wicaksono, I. Šolić, J. Singh, T.S. Kaku, S. Lim, E.W.J. Ang, L.
- Blancafort, T.W.J. Steele, Rapid Activation of Diazirine Biomaterials with the Blue Light
 Photocatalyst, ACS Applied Materials & Interfaces (2021).
- 736 [19] L.J. Macdougall, V.X. Truong, A.P. Dove, Efficient In Situ Nucleophilic Thiol-yne
- 737 Click Chemistry for the Synthesis of Strong Hydrogel Materials with Tunable Properties,
- ACS Macro Letters 6(2) (2017) 93-97.
- 739 [20] C.E. Hoyle, C.N. Bowman, Thiol–ene click chemistry, Angewandte Chemie
- 740 International Edition 49(9) (2010) 1540-1573.
- [21] A.E. Rydholm, C.N. Bowman, K.S. Anseth, Degradable thiol-acrylate photopolymers:
- polymerization and degradation behavior of an in situ forming biomaterial, Biomaterials
 26(22) (2005) 4495-4506.
- [22] P.M. Kharkar, M.S. Rehmann, K.M. Skeens, E. Maverakis, A.M. Kloxin, Thiol-ene
- click hydrogels for therapeutic delivery, ACS biomaterials science & engineering 2(2) (2016)165-179.
- 747 [23] Y. Dong, A.O. Saeed, W. Hassan, C. Keigher, Y. Zheng, H. Tai, A. Pandit, W. Wang,
- 748 "One-step" Preparation of Thiol-Ene Clickable PEG-Based Thermoresponsive
- Hyperbranched Copolymer for In Situ Crosslinking Hybrid Hydrogel, Macromolecular rapid
 communications 33(2) (2012) 120-126.
- 751 [24] K. Jin, N. Wilmot, W.H. Heath, J.M. Torkelson, Phase-Separated Thiol–Epoxy–Acrylate
- 752 Hybrid Polymer Networks with Controlled Cross-Link Density Synthesized by Simultaneous
- 753 Thiol–Acrylate and Thiol–Epoxy Click Reactions, Macromolecules 49(11) (2016) 4115-
- 754 4123.
- 755 [25] I. Djordjevic, G. Wicaksono, I. Solic, T.W. Steele, In Vitro Biocompatibility of
- 756 Diazirine-Grafted Biomaterials, Macromolecular Rapid Communications 41(21) (2020)
- 757 2000235.
- [26] B. Steyrer, P. Neubauer, R. Liska, J. Stampfl, Visible light photoinitiator for 3D-printing
 of tough methacrylate resins, Materials 10(12) (2017) 1445.
- 760 [27] J.C. Worch, C.J. Stubbs, M.J. Price, A.P. Dove, Click Nucleophilic Conjugate Additions
- to Activated Alkynes: Exploring Thiol-yne, Amino-yne, and Hydroxyl-yne Reactions from
- 762 (Bio)Organic to Polymer Chemistry, Chemical Reviews 121(12) (2021) 6744-6776.
- 763 [28] L. Maleki, U. Edlund, A.-C. Albertsson, Synthesis of full interpenetrating hemicellulose
- hydrogel networks, Carbohydrate polymers 170 (2017) 254-263.
- 765 [29] N.B. Cramer, J.P. Scott, C.N. Bowman, Photopolymerizations of thiol– ene polymers
- without photoinitiators, Macromolecules 35(14) (2002) 5361-5365.

- 767 [30] A. Gambi, M. Winnewisser, J.J. Christiansen, The infrared spectrum of diazirine:.
- Rovibrational analysis of the v3 fundamental, Journal of Molecular Spectroscopy 98(2)
 (1983) 413-424.
- [31] R. Cataliotti, A. Poletti, G. Paliani, A. Foffani, Infrared Spectrum and Vibrational
- Assignment of Diazocyclopentadiene-h4 and-d4, Zeitschrift für Naturforschung B 27(8)
- 772 (1972) 875-878.
- [32] M.-A. Tehfe, J. Lalevée, S. Telitel, E. Contal, F.d.r. Dumur, D. Gigmes, D. Bertin, M.
- Nechab, B. Graff, F. Morlet-Savary, Polyaromatic structures as organo-photoinitiator
- catalysts for efficient visible light induced dual radical/cationic photopolymerization and
- interpenetrated polymer networks synthesis, Macromolecules 45(11) (2012) 4454-4460.
- [33] I. Djordjevic, G. Wicaksono, I. Solic, T.W.J. Steele, Diazoalkane decay kinetics from
- UVA-active protein labelling molecules: Trifluoromethyl phenyl diazirines, Results in
 Chemistry 2 (2020) 100066.
- 780 [34] P. Luan, G.S. Oehrlein, Characterization of Ultrathin Polymer Films Using p-Polarized
- ATR-FTIR and Its Comparison with XPS, Langmuir 35(12) (2019) 4270-4277.
- [35] I. Djordjevic, N. Choudhury, N. Dutta, S. Kumar, Poly(octanediol-co-(citric acid)-co-
- (sebacic acid)) elastomers: novel bio-elastomers for tissue engineering, Polymer International
 60 (2011) 333-343.
- [36] I. Djordjevic, N.R. Choudhury, N.K. Dutta, S. Kumar, E.J. Szili, D.A. Steele,
- 786 Polyoctanediol Citrate/Sebacate Bioelastomer Films: Surface Morphology, Chemistry and
- Functionality, Journal of Biomaterials Science, Polymer Edition 21(2) (2010) 237-251.
- [37] O. Türünç, M.A.R. Meier, A novel polymerization approach via thiol-yne addition,
- Journal of Polymer Science Part A: Polymer Chemistry 50(9) (2012) 1689-1695.
- [38] M. Sahin, S. Ayalur-Karunakaran, J. Manhart, M. Wolfahrt, W. Kern, S. Schlögl, Thiol-
- 791 Ene versus Binary Thiol–Acrylate Chemistry: Material Properties and Network
- 792 Characteristics of Photopolymers Advanced Engineering Materials 19(4) (2017) 1600620.
- [39] H. Saraf, K.T. Ramesh, A.M. Lennon, A.C. Merkle, J.C. Roberts, Mechanical properties
- of soft human tissues under dynamic loading, Journal of Biomechanics 40(9) (2007) 1960-1967.
- 796 [40] Q.L. Loh, C. Choong, Three-Dimensional Scaffolds for Tissue Engineering
- 797 Applications: Role of Porosity and Pore Size, Tissue Engineering Part B: Reviews 19(6)
- 798 (2013) 485-502.
- [41] A.B. Kumar, J.D. Tipton, R. Manetsch, 3-Trifluoromethyl-3-aryldiazirine photolabels
- 800 with enhanced ambient light stability, Chemical Communications 52(13) (2016) 2729-2732.
- [42] S.P. Green, K.M. Wheelhouse, A.D. Payne, J.P. Hallett, P.W. Miller, J.A. Bull, Thermal
- 802 Stability and Explosive Hazard Assessment of Diazo Compounds and Diazo Transfer
- Reagents, Organic Process Research & Development 24(1) (2020) 67-84.
- [43] M.L. Lepage, C. Simhadri, C. Liu, M. Takaffoli, L. Bi, B. Crawford, A.S. Milani, J.E.
- Wulff, A broadly applicable cross-linker for aliphatic polymers containing C–H bonds,
 Science 366(6467) (2019) 875-878.
- [44] G.M. Cruise, D.S. Scharp, J.A. Hubbell, Characterization of permeability and network
- structure of interfacially photopolymerized poly(ethylene glycol) diacrylate hydrogels,
 Biomaterials 19(14) (1998) 1287-1294.
- [45] A. Bagheri, J. Jin, Photopolymerization in 3D Printing, ACS Applied Polymer Materials
 1(4) (2019) 593-611.
- 812 [46] W. Li, L.S. Mille, J.A. Robledo, T. Uribe, V. Huerta, Y.S. Zhang, Recent Advances in
- Formulating and Processing Biomaterial Inks for Vat Polymerization-Based 3D Printing,
- Advanced Healthcare Materials 9(15) (2020) 2000156.

- [47] W.J. Steele, Terry ; Djordjevic, I., Hygroscopic, Crosslinking Coatings and
- Bioadhesives, <u>https://patentscope.wipo.int/search/en/detail.jsf?docId=WO2019050479</u>
 PCT/SG2018/050452 (2019).
- [48] M. Singh, C.S. Yin, S.J. Page, Y. Liu, G. Wicaksono, R. Pujar, S.K. Choudhary, G.U.
- 819 Kulkarni, J. Chen, J.V. Hanna, R.D. Webster, T.W.J. Steele, Synergistic Voltaglue Adhesive
- Mechanisms with Alternating Electric Fields, Chemistry of Materials 32(6) (2020) 24402449.
- [49] J. Ping, F. Gao, J.L. Chen, R.D. Webster, T.W. Steele, Adhesive curing through low-
- voltage activation, Nature communications 6(1) (2015) 1-9.
- [50] G. Feng, I. Djordjevic, V. Mogal, R. O'Rorke, O. Pokholenko, T.W.J. Steele, Elastic
- Light Tunable Tissue Adhesive Dendrimers, Macromolecular Bioscience 16(7) (2016) 10721082.
- [51] A. Fu, K. Gwon, M. Kim, G. Tae, J.A. Kornfield, Visible-light-initiated thiol–acrylate
- photopolymerization of heparin-based hydrogels, Biomacromolecules 16(2) (2015) 497-506.
- [52] R. Taschner, P. Gauss, P. Knaack, R. Liska, Biocompatible photoinitiators based on
- poly- α -ketoesters, Journal of Polymer Science 58(2) (2020) 242-253.