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Du, Guangsheng ; Zhang, Zhihua; He, Penghui ; Zhang, Zhibing; Sun. Xun

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1 Determination of the mechanical properties of polymeric microneedles by micromanipulation

2 Guangsheng Du¹, Zhihua Zhang², Penghui He¹, Zhibing Zhang^{2*} and Xun Sun^{1*}

³ ¹West China School of Pharmacy, Sichuan University, Chengdu, 610041, China

- ² School of Chemical Engineering, University of Birmingham, Edgbaston, Birmingham B15 2TT, UK
- 5

6 *Corresponding author

7 E-mail address: <u>sunxun@scu.edu.cn</u>; <u>Z.Zhang@bham.ac.uk</u>

8 Abstract:

9 Precise characterization of the mechanical properties of polymeric microneedles is crucial for their 10 successful penetration into skin and delivery of the loaded active ingredients. However, most available 11 strategies for this purpose are based on compression of the whole patch, which only provide the 12 average rupture force of the needles and can not give information on the variations across individual 13 microneedles in the patch. In this study, we determined the mechanical strength of individual 14 microneedles of two types of hyaluronic acid microneedles with or without loaded model drugs using a 15 micromanipulation technique. The applied force as a function of displacement of the microneedles was 16 recorded, which was used to determine the rupture displacement, rupture force, and then to derive and calculate normal stress-deformation curve, rupture stress and Young's modulus of individual 17 18 microneedles. The obtained data suggest that the molecular weight of the polymer and the loading of 19 drug into the microneedles can significantly affect the rupture behavior and mechanical properties of 20 the microneedles, which provides a foundation for preparing sufficiently strong microneedles for 21 controlled drug delivery.

Keywords: Mechanical properties, Polymeric microneedles, Micromanipulation, Rupture force, Normalstress

24 1. Introduction

25 Polymeric microneedles have been widely investigated for drug delivery, medical diagnosis and health 26 monitoring [1-3]. They can pierce skin barrier in a non-invasive and pain-free way as they do not touch 27 nerves or blood capillaries inside the skin during application. As compared to solid and hollow 28 microneedles made of glass or metal, polymeric microneedles made of dissolving or biodegradable 29 polymers also hold advantages including resulting in no hazardous waste after administration and 30 easiness for modulation of release properties of the loaded ingredients [4,5]. However, polymer based 31 microneedles have a relatively weak mechanical strength, which may cause the breakage or bending of 32 the microneedles during the insertion of skin, resulting in an insufficient penetration [6,7]. Direct and 33 precise measurement of the mechanical properties of polymeric microneedles is necessary for ensuring 34 their successful application especially in case of industrial mass production. Besides, emerging new types 35 of advanced microneedles have been designed and investigated in the past decade, such as bio-36 responsive microneedles [8,9], core-shell structured microneedles [10,11], and hydrogel microneedles

[12]. Precise characterization of the mechanical properties of these microneedles is crucial for their
 possible translation as the complex composition and design could significantly affect their mechanical
 strength.

40 In many microneedle studies, the mechanical strength of microneedle patch was investigated by 41 compressing the whole patch against a flat surface, after which the rupture force of single microneedles 42 was calculated by dividing the total rupture force by the number of needles [7,13]. This strategy is not 43 adequate since it can not identify the possible variations of the mechanical properties among the 44 microneedles across the patch. The mechanical properties obtained are also limited to the rupture force 45 of the bulk patch. In other microneedle studies, their mechanical properties were not directly measured 46 but were instead reflected by their skin penetration efficiency. The small holes in the skin generated by 47 microneedle penetration were normally stained and visualized for calculating the skin penetration 48 efficiency [14,15]. However, this method gives no quantitative results for the mechanical property of 49 microneedles. Atomic force microscopy (AFM) has also been used to measure the mechanical properties 50 of microneedles. However, their indenting depth or force is limited to nanoscale measurement [16,17].

51 Micromanipulation is an experimental technique that was first developed for measurement of the 52 bursting force of single mammalian cells [18]. This technique was then extended to analyze other 53 biological or non-biological micro-particles, including microcapsules and microspheres [19-21]. Micro-54 particles can be rested on a glass slide or in a chamber and then compressed by using a cylindrical probe 55 with a diameter larger than the micro-particles. As compared to other bulk methods for mechanical 56 property characterization, the technique can record the force-displacement curves generated from 57 compression of single micro-particles, which can be further used to extract important mechanical 58 property parameters including rupture force, displacement at rupture and normal rupture stress [22,23]. 59 The generated data can also be used to determine the intrinsic material properties of the samples, such 60 as Young's modulus, yield stress and stress and strain at rupture by analytical or numerical modeling 61 [19].

62 Herein, we investigated the mechanical properties of two types of home-made hyaluronic acid (HA) 63 based dissolving microneedles with or without loaded model drugs of lidocaine hydrochloride and 64 bupivacaine hydrochloride with a micromanipulation technique. Both of them are hydrophilic small-65 molecule drugs and used as local anesthetic of the amino amide type in clinic. The use of 66 micromanipulation allows the precise measurement of the mechanical properties of individual 67 microneedles. The force-displacement data were first recorded with this technique, which were used to 68 determine the rupture displacement and rupture force, and then to derive normal rupture stress and 69 Young's modulus of the microneedles. The influences of molecular weight of HA and the drug loading on 70 the mechanical properties of the microneedles were investigated.

71 2. Method and materials

72 Materials

HA with a molecular weight of 10-kDa and 300-kDa were purchased from Bloomage Biotech (Jinan, China). Lidocaine hydrochloride and bupivacaine hydrochloride as model drugs were received as gifts from West China Hospital of Sichuan University (Chengdu, China). Rhodamine B dye was purchased from
 Meilunbio (Dalian, China). Milli-Q water was used for the preparation of all solutions. All the other

77 reagents used were of analytical grade.

78 Fabrication of HA based microneedles

79 Different types of fabricated quadrangular pyramid microneedles (700 µm height, 300 µm base width 80 and 600 μ m center-center spacing) in a 10 by 10 array on a back plate of 0.9 \times 0.9 cm² are summarized 81 in Table 1. All of the microneedles were fabricated by using a micro-molding method as previously 82 reported [24]. Briefly, polydimethylsiloxane (PDMS) molds were first duplicated from stainless steel 83 microneedle molds. Next, 30 µl HA solution dissolved with or without drugs was loaded into the cavities 84 of a PDMS mold by using pressurized air to form the needles of the patch. After drying for 0.5 h in 85 anhydrous silica gel environment, 40 µl of blank HA solution was added into the mold to form the back 86 plate of the patch. Finally, the microneedle patches were dried for another 4 h and subsequently peeled 87 off from the mold. The fabricated patches were stored in dry environment for further characterization. 88 To fabricate microneedle patches for visualization by confocal microscopy, rhodamine B dye was 89 dissolved in HA solution for the fabrication.

90	Table 1. Different types of microneedle (MN)	patches characterized in the current study
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MN type	Matrix	Abbreviation name
1	10-kDa HA	Blank HA (10-kDa) MN
2	300-kDa HA	Blank HA (300-kDa) MN
3	Bupivacaine + 10-kDa HA (Weight ratio 1:2)	Bupi HA (10-kDa) MN
4	Lidocaine + 300-kDa HA (Weight ratio 4:5)	Lido HA (300-kDa) MN

91

92 Visualization of microneedles

93 The surface morphology of the fabricated microneedles was visualized using scanning electron 94 microscopy (SEM) (JEOL JSM-7500F, Japan) with an operation voltage of 15 kV. The microneedle patches 95 were fixed on SEM stub and coated with a thin layer of carbon for the visualization. The microneedles 96 containing rhodamine B were visualized by confocal laser scanning microscopy (ZEISS CLSM, Overkochen, 97 Germany) by scanning them with a depth resolution of 5 µm/step using a 4× Plan Apo objective with a 98 4× magnification.

99 Micromanipulation

100 The detailed working principles of the micromanipulation technique have been introduced in previous 101 studies [18,21]. Briefly, to start the compression, the microneedle patches were placed on the stage of

102 the micromanipulation instrument equipped with a force transducer (Model GSO-10, Transducer

103 Techniques, LLC, USA). An optical glass rod made of Borosilicate with a flat end of a diameter of 100 μm

104 was mounted onto the force transducer for the compression. Single microneedles were compressed 105 between the stage and the glass probe with a compression speed of 2 µm/s. The applied force on single 106 microneedles as a function of their displacement was recorded. One typical example of different status 107 of the single microneedles during compression is shown in Figure 1. The compression process is shown 108 in supplementary video. For each type of microneedles, 30 needles were measured.

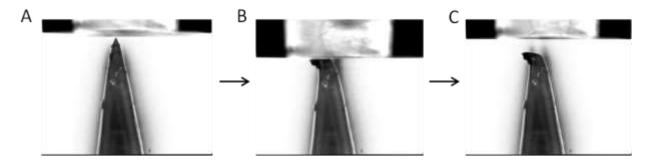


Figure 1. One example of different status of individual microneedle during the compression measurement. A:
 Before compression, B: Fracture of microneedles, C: After compression.

112 Calculation of the normal stress from compression force

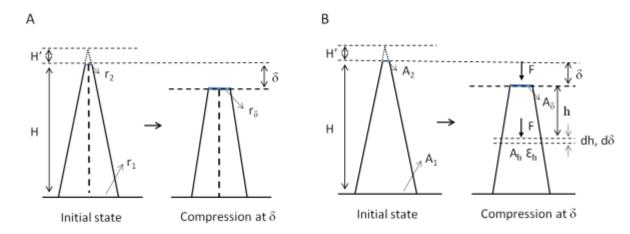


Figure 2. Schematic diagram for modeling of stress (A) and Young's modulus (B). Abbreviations: H: initial 114 115 height of a microneedle. H': height of the missing tip when assuming the microneedle is perfectly sharp. r_1 : half side length of a quadrangular microneedle base. r_2 : initial half side length of the quadrangular 116 microneedle tip. δ : displacement of the microneedle. r_{δ} : half side length of contact surface at a 117 118 compressing displacement of δ . A₁: surface area of the quadrangular microneedle base. A₂: surface area 119 of the quadrangular microneedle tip. A_{δ} : contact surface area at a compressing displacement of δ . F: 120 compression force at a displacement of δ . h: height of the microneedle (from top of the microneedle) 121 until sectional area of A_h. \mathcal{E}_h : deformation at a height of h. dh: the length of the element at h. d δ : 122 displacement at h.

123

113

109

124 The normal stress was calculated from compression force using modeling as shown in Figure 2A. In this 125 modeling, we ignored the volume change of microneedle and the deformation of microneedle base 126 during compression (Figure 2A).

127 The volume of a single microneedle can be calculated by using dimension parameters as below (Eq.1).

128
$$V=4/3r_1^2(H+H')-4/3r_2^2H'$$
 (Eq.1)

where V is the volume of the microneedle, r_1 is half side length of the quadrangular microneedle base, H is initial height of the microneedle, H' is height of the missing tip when assuming the microneedle is

131 perfectly sharp and r_2 is half initial side length of the quadrangular microneedle tip.

132 According to geometric similarity (Eq.2)

133
$$H'/r_2 = (H+H')/r_1$$
 (Eq.2)

By combining microneedle volume (Eq.1) and geometric similarity (Eq.2) equations, we can eliminate H' from the equations, and r_2 can be calculated from parameters which are already known (Eq.3).

- 136 $r_2 = 1/2(-r_1 + \sqrt{3V/H 3r_1^2})$ (Eq.3)
- 137 After replacing r_2 with r_δ and replacing H by H- δ at a compression displacement of δ , r_δ can be obtained 138 (Eq.4).
- 139 $r_{\delta} = 1/2(-r_1 + \sqrt{3V/(H \delta) 3r_1^2})$ (Eq.4)
- Finally, the normal stress can be calculated by dividing compression force by the contacting area 4 r_{δ}^2 (Eq. 5).
- 142 Normal stress = $F/(4r_{\delta}^2)$ (Eq.5)
- 143

144 Young's modulus calculation

145 Young's modulus was calculated from compression force using a modeling as shown in Figure 2B.

146 We assumed that the microneedle was linearly elastic under a small deformation and only considered

147 the longitudinal deformation. At a height of h (from the top of the microneedle) with the sectional area

148 A_h , the deformation ε_h can be calculated by the stress-strain equation (Eq.6).

$$\epsilon_{h} = \frac{F}{EA_{h}}$$
 (Eq. 6)

- 149 Where F is the compression force, E is Young's modulus.
- 150 ϵ_h can also be given by

$$\epsilon_{\rm h} = \frac{{\rm d}\delta}{{\rm d}{\rm h}}$$
(Eq.7)

151 where $d\delta$ is the displacement at h and dh is the length of the element at h.

152 From geometric similarity

$$\frac{A_{h}}{A_{1}} = \left(\frac{h + H'}{H + H'}\right)^{2}$$
 (Eq. 8)

153 Thus,

$$A_{h} = \left(\frac{h + H'}{H + H'}\right)^{2} A_{1}$$
 (Eq.9)

154 The overall displacement δ can be calculated by integrating $d\delta$ over the microneedle as

$$\delta = \int_0^H d\delta \tag{Eq.10}$$

155 Submitting Eq.6 to 9 into Eq.10 leads to

$$\delta = \frac{F}{A_{1}E} \int_{0}^{H} \frac{1}{\left(\frac{h + H'}{H + H'}\right)^{2}} dh$$
 (Eq.11)

156 which can be written as follows

$$E = \frac{F}{\delta} \frac{(H+H')H}{A_1H'}$$
(Eq.12)

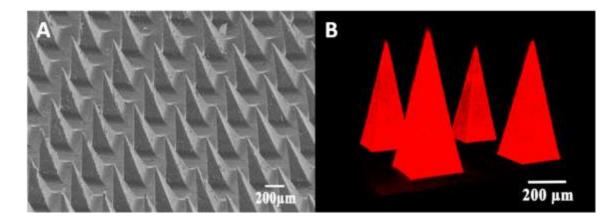
157 Statistical analysis

158 In total 30 microneedles from each type of samples were analyzed. All the data of the mechanical 159 properties were analyzed by one way ANOVA with Turkey's post-test using GraphPad Prism software

160 (version 5.02). The significance levels were set at *p<0.05, **p<0.01, and ***p<0.001.

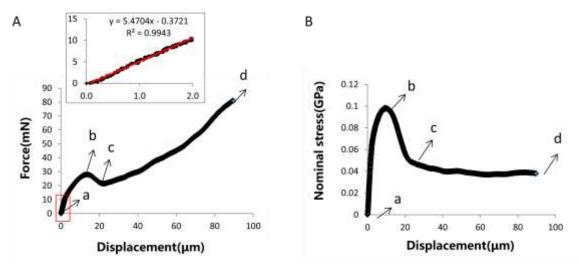
161 **3. Results**

162 Morphology of microneedles



- Figure 3. Representative microscopy images of the fabricated microneedles. A: Scanning electronic microscopy
 (SEM) images. B: Confocal laser scanning microscopy (CLSM) images.
- 166 The microscopy images of representative microneedles in a patch are shown in Figure 3. SEM (Figure 3A)
- and CLSM (Figure 3B) images show that the obtained microneedles had a regular quadrangular pyramid
- 168 shape and sharp tips.



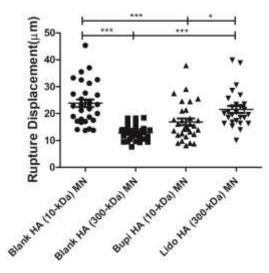


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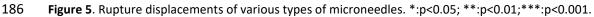
Figure 4. Typical force versus displacement (A) and the corresponding normal stress versus displacement (B) curvesobtained from compression of a single microneedle.

173 A typical relationship of the force applied on an individual microneedle and the corresponding normal 174 stress as a function of the probe moving distance (displacement) is shown in Figure 4A and 4B, 175 respectively. Immediately after point-a the probe touched the microneedle tip, the force increased until 176 point-b at which the microneedle ruptured (Figure 4A). As a result, the force decreased to point-c and 177 increased again until reaching the detection limitation of the sensor at point-d. Figure 4B shows the 178 derived normal stress-displacement curve. In contrast, the normal stress increased from point-a to peak 179 point-b when the needle ruptured, after which the stress decreased to point-c and reached a plateau 180 where it stayed roughly the same until the end of the measurement (point-d). From these curves, the 181 rupture force and rupture stress (normal stress at rupture) can be determined (point-b). The rupture 182 displacement is identified as the distance that the probe travels from point-a until point-b when the 183 microneedle ruptures. The video of the compression process is shown in Supplementary information.

184 Summary of the rupture displacements



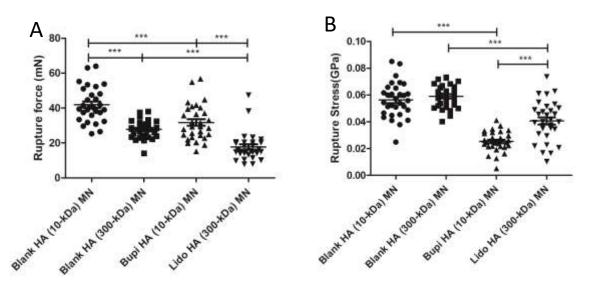




187 We summarized the rupture displacements of different types of microneedles in Figure 5. It is shown that Blank HA (300-kDa) MN patch had the narrowest distribution of rupture displacement among all 188 189 patches. Specifically, the characterized patches showed an average rupture displacement between 12.9 190 \pm 2.8 µm (Blank HA (300-kDa) MN) and 23.8 \pm 7.9 µm (Blank HA (10-kDa) MN). The rupture displacement 191 of microneedles made of 10-kDa HA (23.8 \pm 7.9 μ m) was significantly higher than that of microneedles 192 made of 300-kDa HA (12.9 \pm 2.8 μ m). The drug loaded patch showed either a lower (Bupi HA (10-kDa) 193 MN, 16.9 \pm 6.9 μ m) or higher (Lido HA (300-kDa) MN, 21.6 \pm 6.8 μ m) rupture displacement than the

194 corresponding blank HA patches.

195 Summary of rupture forces and stresses

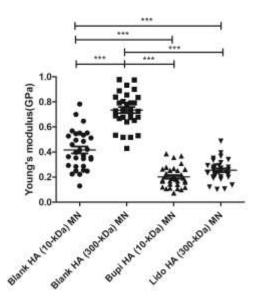


196

197 Figure 6. Rupture forces and stresses of different types of microneedles. *:p<0.05; **:p<0.01; ***:p<0.001.

The rupture forces and stresses of different types of microneedle patches are summarized and 198 199 presented in Figure 6. In case of rupture force, blank microneedles made of 10-kDa HA showed a mean 200 rupture force of 42.0 ± 9.9 mN, which was significantly higher than that of the microneedles made of 201 300-kDa HA (27.7 ± 5.2 mN). The loading of drug in HA (10-kDa) MN and HA (300-kDa) MN significantly 202 decreased the rupture force of the corresponding microneedles to 31.7 ± 10.3 mN and 17.7 ± 8.3 mN, 203 respectively. As for rupture stress, by contrast, blank microneedles made of 10-kDa and 300-kDa HA 204 showed similar rupture stresses of 0.072 \pm 0.017 GPa and 0.075 \pm 0.011 GPa. The loading of drug into 205 the microneedles made of 10-kDa and 300-kDa HA significantly decreased the rupture stresses by 206 around 50% and 25%, respectively.

207 Young's modulus of the microneedles



208

Figure 7. Calculated Young's modulus of different types of microneedles. *:p<0.05; **:p<0.01;***:p<0.001

210 The relationship between force and displacement was found be linear for displacements up to $2 \mu m$, as 211 predicted by the elastic model (Eq.9), which was used to calculate the Young's modulus of different 212 microneedles via liner regression (see insert of Figure 4A) and the Young's modulus values are presented 213 in Figure 7. Interestingly, blank microneedles made of 300-kDa HA showed a significantly higher Young's 214 modulus than that made of 10-kDa HA. The incorporation of drugs into microneedles also significantly 215 decreased the Young's modulus of the microneedles: the loading of bupivacaine into 10-kDa HA MN 216 slightly decreased the Young's modulus from 0.42 \pm 0.15 to 0.20 \pm 0.08 GPa while the loading of 217 lidocaine into 300-kDa HA microneedles robustly decreased the Young's modulus from 0.73 \pm 0.14 to 218 0.25 ± 0.09 GPa.

219 4. Discussion and conclusions

220 One aim of this study was to investigate the potential of the micromanipulation technique for 221 characterization of the mechanical properties of polymeric microneedles. We therefore fabricated HA 222 microneedles with two different molecular weights with or without loaded model drugs. We measured 223 and compared their mechanical properties with micromanipulation. Our results showed that the micro-224 manipulation technique is a powerful tool for precise and comprehensive characterization of the 225 mechanical properties of polymeric microneedles. The change of the compression force as a function of 226 displacement behavior of single microneedles can be precisely recorded during the compression. With 227 this information as well as mathematic modeling, the normal stress at the contact area between the 228 force probe and the microneedle as a function of its displacement and several important 229 mechanical/material property parameters including displacement at rupture, rupture force and Young's 230 modulus can be analyzed and calculated. As compared to bulk compression strategies, 231 micromanipulation has at least two advantages. Firstly, in comparison with different bulk compression 232 strategies as previously reported, which assume all microneedles within one patch have the same 233 mechanical strength, micro-manipulation can directly and precisely measure the rupture behavior of 234 individual microneedles, therefore can give information on the uniformity of the microneedle strength 235 across the patch. Our data indeed showed that the mechanical property parameters of microneedles 236 across one patch can vary significantly (Figure 5-7). Secondly, the bulk compression strategies only 237 measure rupture forces to evaluate the mechanical property of microneedle patches. Additionally, we 238 compressed the whole patch of Blank HA (10-KDa) MN with a traditional strategy. We even could not 239 observe a clear rupture behavior of the patch (Supplementary Figure 1). In micro-manipulation, as 240 shown in our results, other parameters including displacement at rupture, rupture stress as well as 241 intrinsic material properties such as Young's modulus can also be measured. Therefore, the 242 micromanipulation technique can be used to comprehensively evaluate the mechanical properties of 243 microneedles within one patch and to compare the difference in mechanical properties among different 244 type of patches.

245 During the measurement, we obtained in total four mechanical/material property parameters that can 246 reflect the mechanical strength of microneedles, including displacement at rupture, rupture force, 247 rupture stress and Young's modulus. Young's modulus can reflect the stiffness of the measured sample, 248 while displacement at rupture describes the maximum deformation level before the rupture of 249 microneedles [25,26]. Both of these two indicators can be used for comparison of the mechanical 250 strength of different microneedles when the measured microneedles are made of the same materials. 251 This is indicated by our results that although Blank HA (10-kDa) MN showed a lower Young's modulus 252 (Figure 7), they showed a higher rupture force as compared to the Blank HA (300-kDa) MN (Figure 6). On the other hand, Blank HA (10-kDa) MN showed a much higher rupture displacement than Blank HA (300-253 254 kDa) MN (Figure 5), but they own a similar rupture stress to the later one (Figure 6).

255 Most of the reported studies made use of rupture force of the microneedles for evaluation of the 256 mechanical strength of microneedles [8,10,27,28]. However, our results showed that rupture force and 257 rupture stress results could give different trends. For example, based on rupture force results, Blank HA 258 (10-kDa) MN is mechanically stronger than Blank HA (300-kDa) MN (Figure 6A). However, their 259 mechanical strengths are similar according to rupture stress results (Figure 6B). Actually, the rupture 260 force of microneedles could be significantly affected by the rupture displacement/contacting area at 261 rupture. As a result, simple comparison of rupture force is not enough for the evaluation of the 262 mechanical strength of different microneedle patches since it is not an intrinsic material property parameter. In contrast, the rupture stress of microneedles can be considered to be an intrinsic material parameter, and our results indicated that it is a better indicator for the mechanical strength of polymer microneedles as compared to rupture force. The rupture stress results in the current study indicated that among the different type of microneedles that were investigated, Blank HA (10-kDa) MN and Blank HA (300-kDa) MN have a similar mechanical strength, but are all stronger than their corresponding drug loaded microneedle patches.

269 Our results in this study showed that both the molecular weight of the polymer and the loading of drug 270 could significantly influence the mechanical properties of microneedles. We showed that blank microneedles made of 300-kDa HA had a significantly higher Young's modulus than that made of 10-kDa 271 272 HA. This observed trend is consistent with a previous study showing that 50 kDa HA gel had a 273 significantly higher elastic modulus than 10-kDa HA gel [29]. However, as discussed above, the greater 274 Young's modulus of Blank HA (300-kDa) MN did not simply lead to a greater rupture stress because it 275 was ruptured at smaller displacement/deformation. Sun et al. also [28] showed that the tensile strength 276 of a material was proportional to Young's modulus for a given material, but the relationship can vary 277 significantly with materials. On the other hand, our results showed that the loading of both lidocaine 278 hydrochloride and bupivacaine hydrochloride significantly decreased the mechanical strength of the 279 microneedles, as shown by the results of rupture force, rupture stress and Young's modulus of the 280 microneedles (Figure 6-7). These results are consistent with previous reports. For instance, Park et al. 281 [13] showed that the loading of calcein or bovine serum albumin into poly-lactide-co-glycolic (PLGA) 282 based microneedles could significantly decrease the mechanical strength of the microneedles. These 283 findings indicated that the mechanical strength of the microneedles was mainly contributed by the 284 polymer matrix and the poor adhesion between the drug and polymer could cause mechanical failure 285 sites for the microneedles [13]. Nevertheless, previous research has indicated that the ultimate rupture 286 stress of skin was around 0.027 ± 0.009 GPa. Our measured rupture stress results of microneedle 287 patches except Bupi HA (10-kDa) are higher than this value, indicating that these microneedles may be 288 strong enough for piercing of the skin [30,31]. It should be noted that the environmental factors 289 including temperature and air humidity could significantly affect the mechanical property of HA 290 microneedles. For instance, Wang et al. has observed that after storing HA microneedles in 60% relative 291 humidity for 30 min at 25 °C, the mechanical strength of the microneedles became significantly lower 292 [32]. The environmental factors will be closely monitored in our future measurements.

In conclusion, our studies showed that the micromanipulation technique is an effective tool for precise characterization of the mechanical properties of polymeric microneedles. The generated information could provide important information for rational design of polymeric microneedles with sufficient mechanical strength for skin penetration. The results also revealed that we should carefully evaluate the variations among the microneedles across the patches and pay attention to the influence of drug loading and molecular weight of the polymer microneedles on their mechanical strength.

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- 304 Conflict of interest
- 305 There is no conflict of interest.
- 306 References:
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